

Macro-, Micro-, and Nanoscale Structures

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how to model in a correct manner within Continuum Mechanics

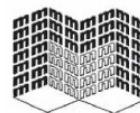
Holm Altenbach

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Krasnoyarsk, Russian Federation



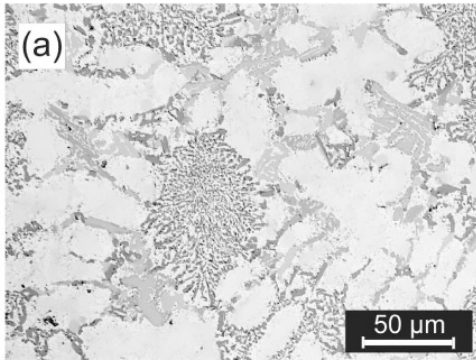
FAKULTÄT FÜR
MASCHINENBAU



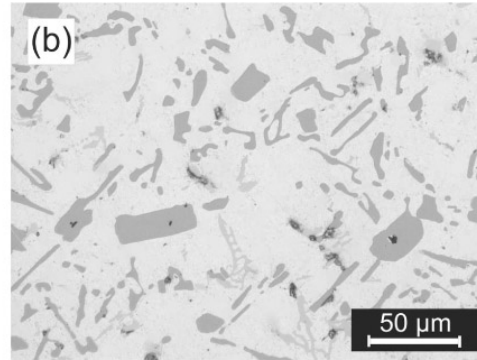
DFG - Graduiertenkolleg
Micro-Macro-Interactions
of Structured Media and Particle Systems

Sr-modified Alloy

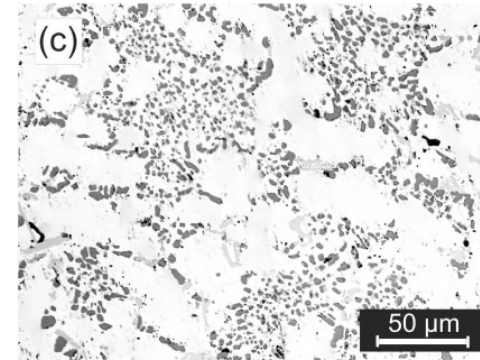
Microstructure (light microscope observation)



M-F alloy (a)



UM-T6 alloy (b)



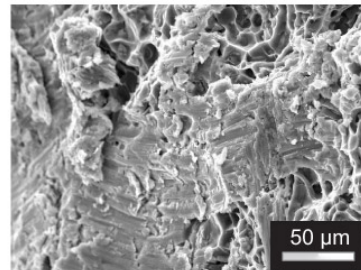
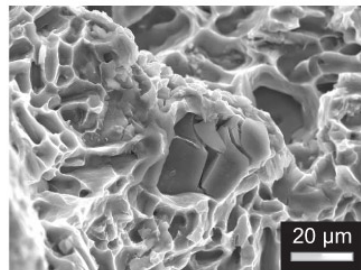
M-T6 alloy (c)

Ultimate tensile strength (UTS), yield strength (YS), and ultimate elongation (UE) for investigated alloys

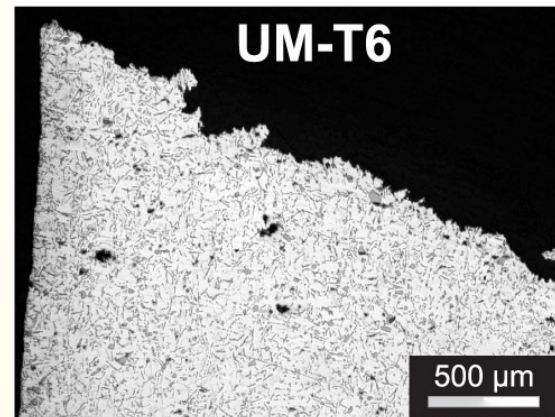
Alloy code	at 20 ⁰ C			at 300 ⁰ C		
	YS, MPa	UTS, MPa	UE, %	YS, MPa	UTS, MPa	UE, %
M-F	132	208	1.0	99	110	5.6
UM-T6	350	358	0.3	136	149	2.9
M-T6	341	363	0.8	138	143	4.8

Fractography

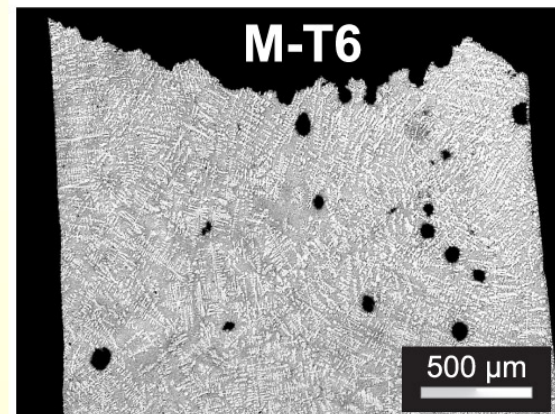
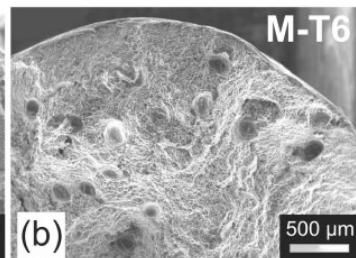
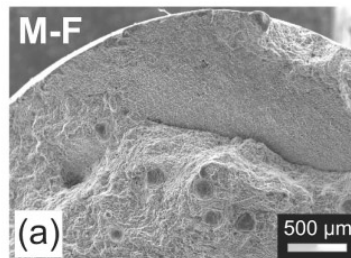
UM-T6 alloy



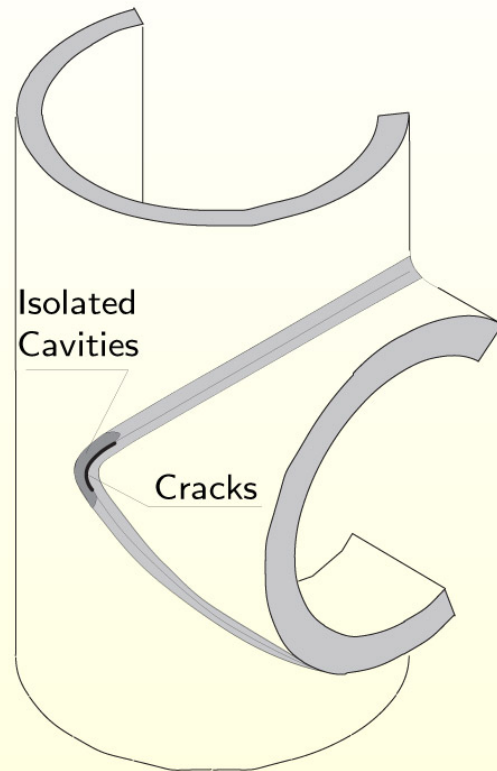
Longitudinal section



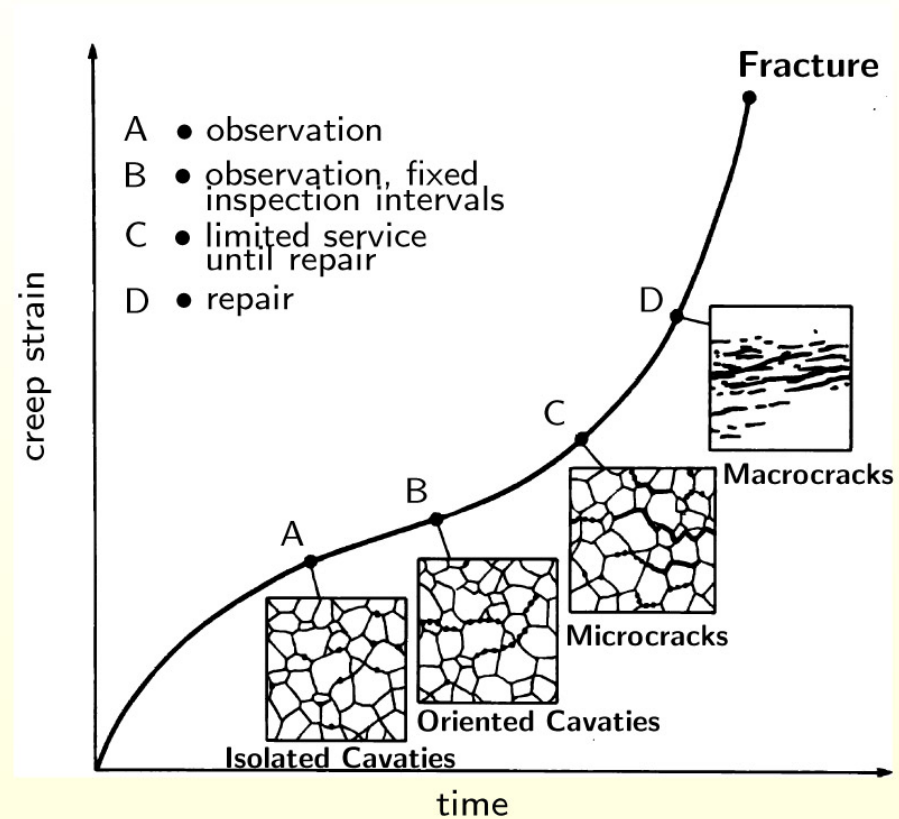
M-F (a) and M-T6 (b) alloys



Welded Structure and Life Cycle



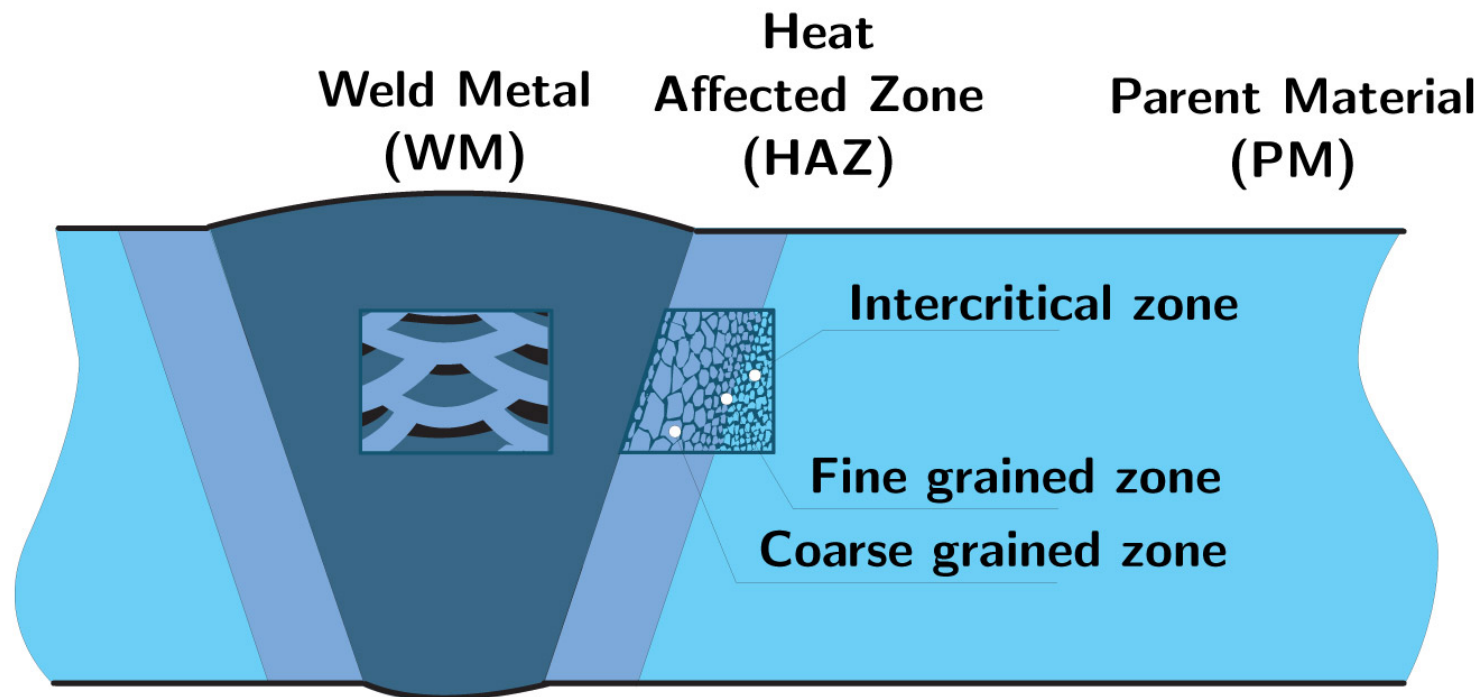
Residual life assessment in a weld region
(after Cella & Fossati, 1992)



Classification of creep damage and corresponding action
(after Neubauer & Wedel, 1983)

Microstructure of Welds

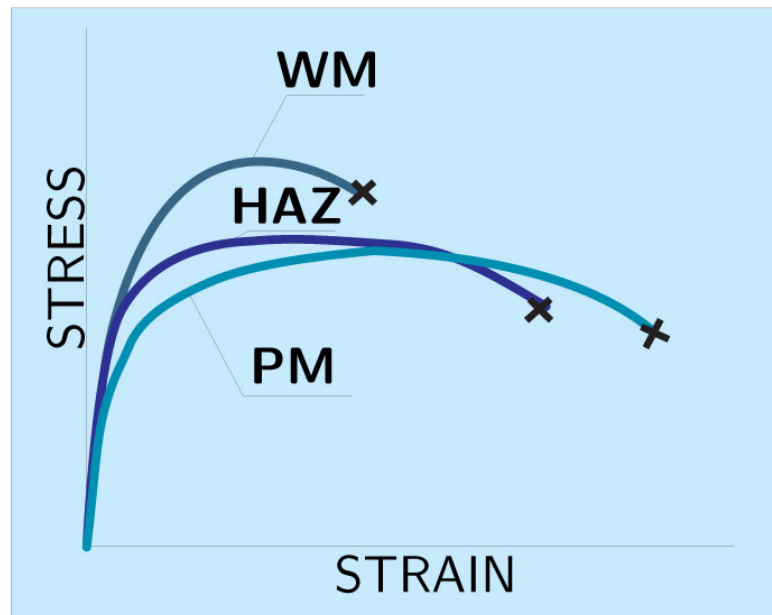
after *Easterling* (1992), *Hyde et al.* (2003), *Wohlfahrt et al.* (2001)



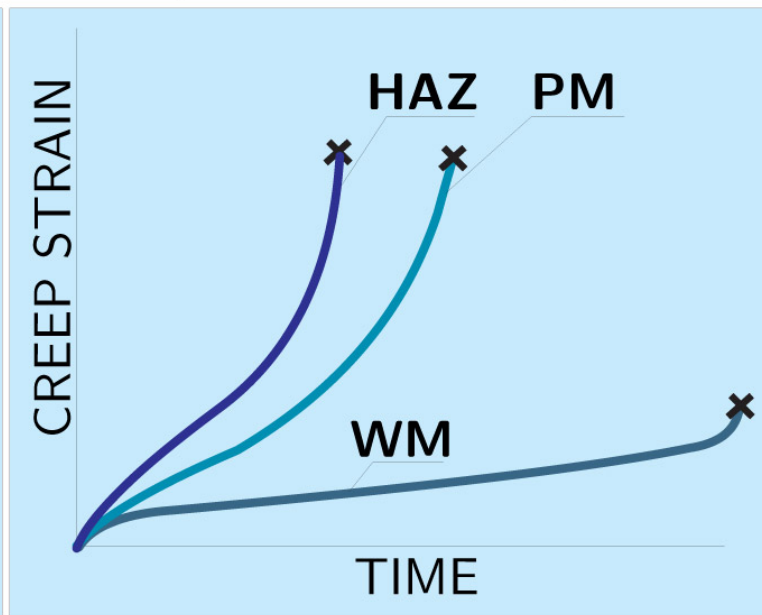
Creep Behavior

after *Easterling* (1992), *Hyde et al.* (2003), *Wohlfahrt et al.* (2001)

Stress-strain behavior
(room temperature)



Creep behavior
($T = 0.5 - 0.7 T_m$)

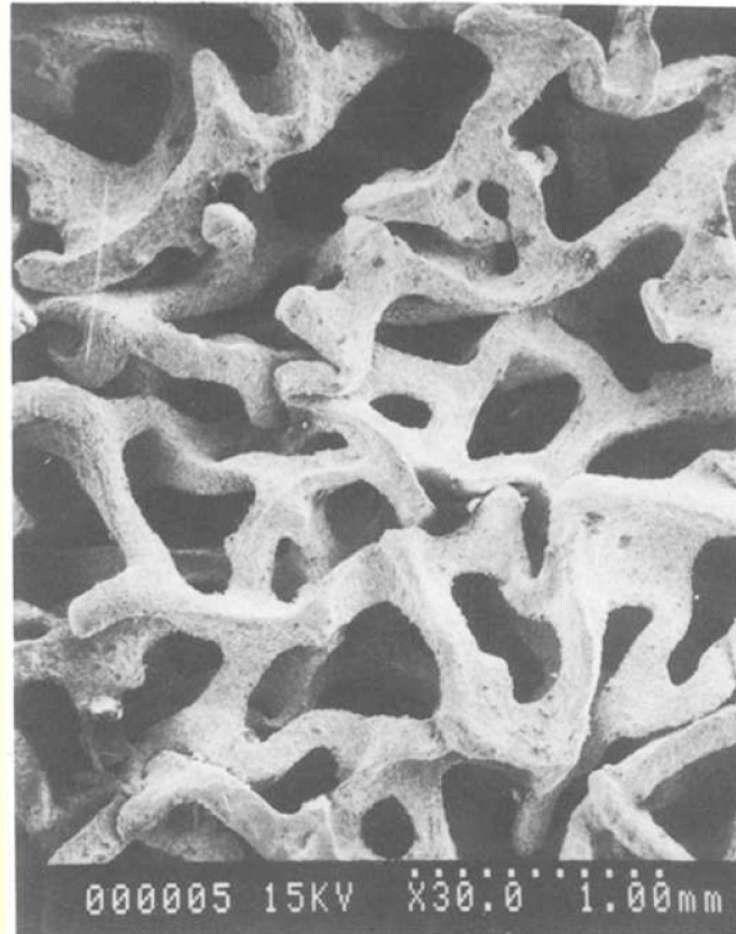
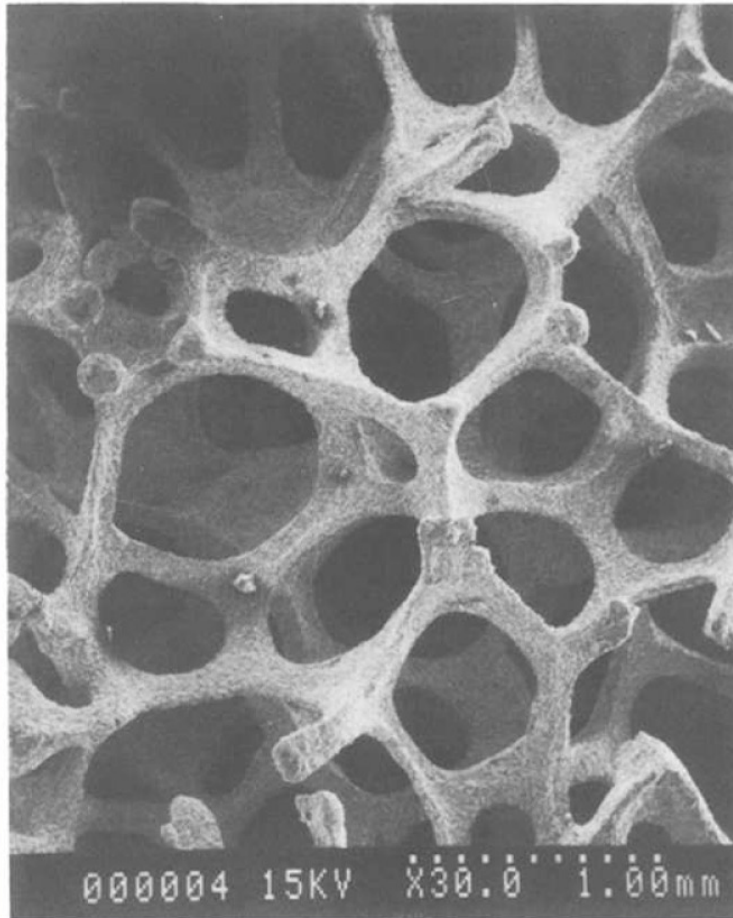


Foams as Nonhomogeneous Materials

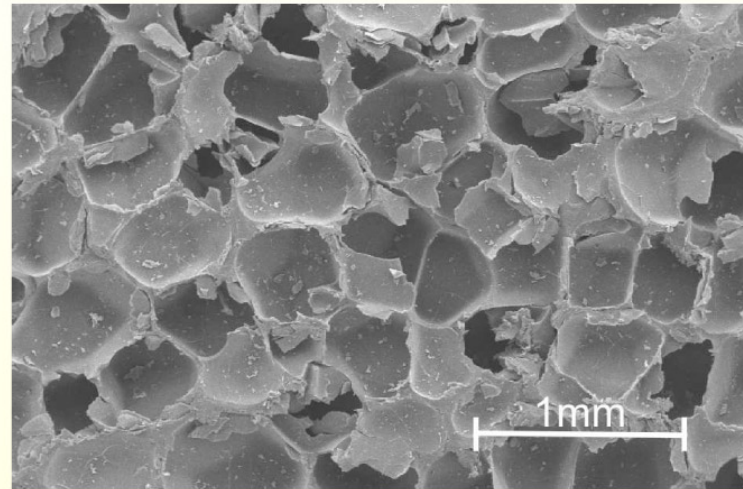
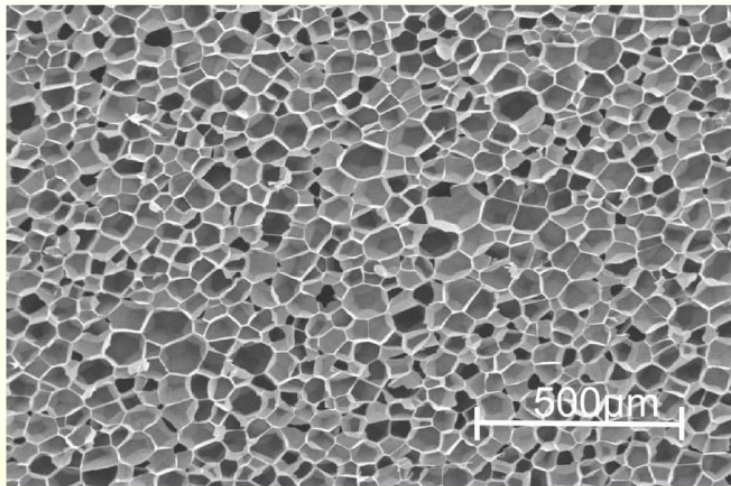
- All materials show a more or less significant microstructure. In many cases the microstructure can be ignored in problems of structural mechanics.
- In the case of foams or porous materials the microstructure must be taken into account. For the first analysis of the microstructure properties can be averaged, e.g. introducing distribution laws.
- Engineering structures made of **foams**, have been used in different applications.¹ **Polymer foam** is a cellular structure consisting of a solid polymer, for example polyurethane, etc., containing a large volume fraction of gas-filled pores. The defining property is the **very high porosity**: typically well over 80%, 90% and even 98% of the volume consists of void spaces.

¹Gibson L J, Ashby M F (1997) Cellular Solids: Structure and Properties;
Mills N (2007) Polymer Foams Handbook Engineering and Biomechanics
Applications and Design Guide

Copper Foams



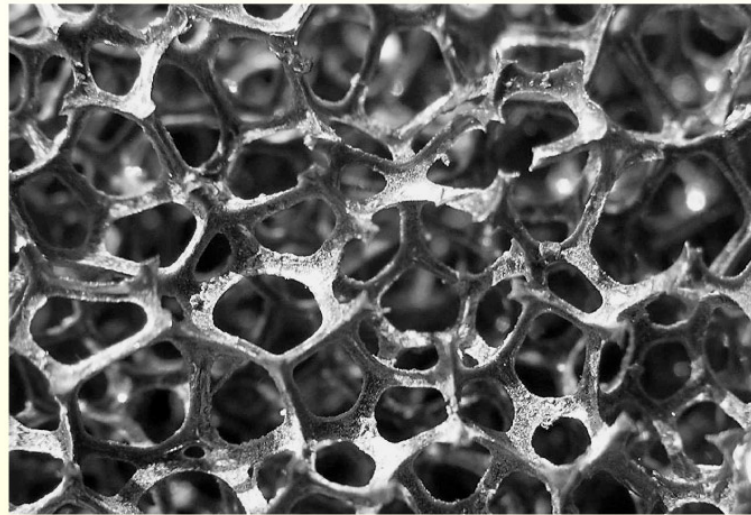
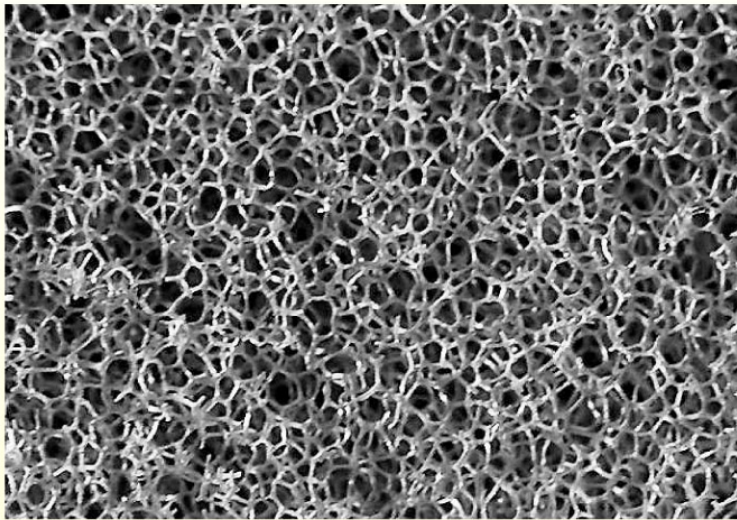
Closed-cell Foams²



Closed-cell polymeric foams with various densities

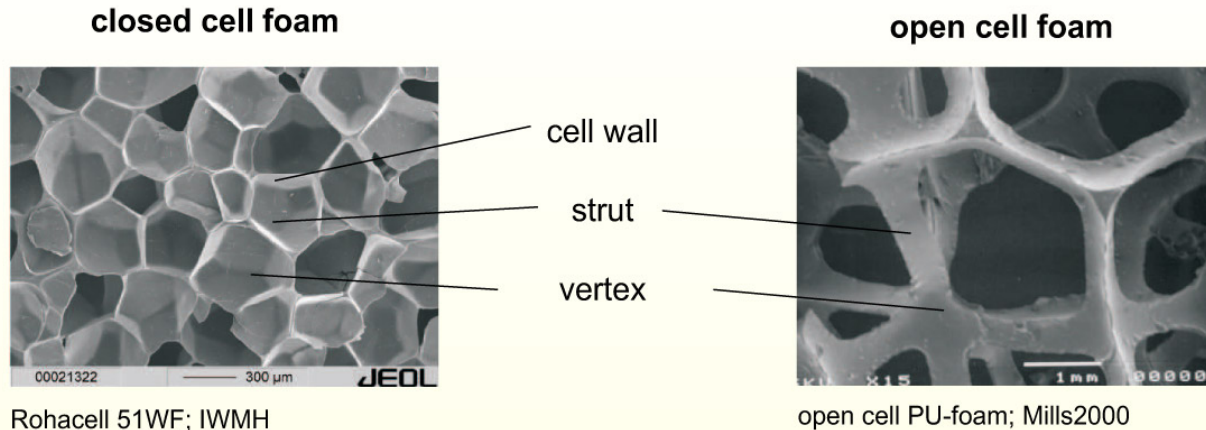
²Kraatz (2007), DKI, Darmstadt, Germany

Open-cell Foams



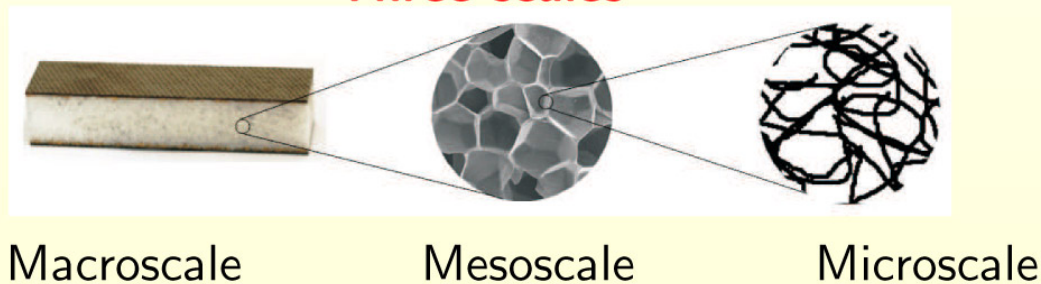
Open-cell foams with various densities

Open and Closed Cell Foams

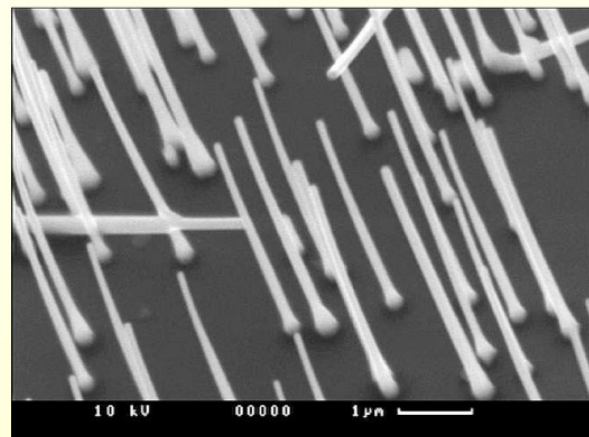
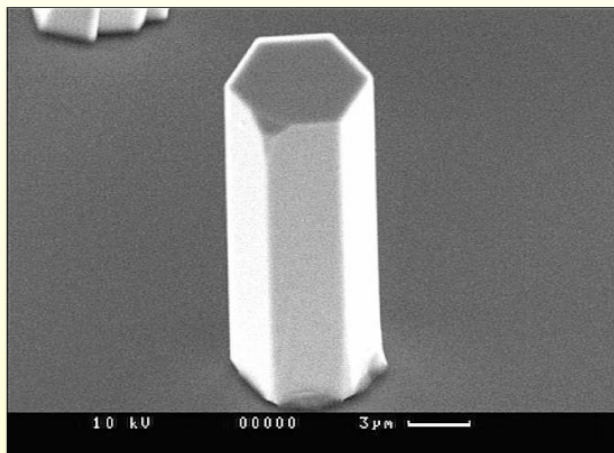
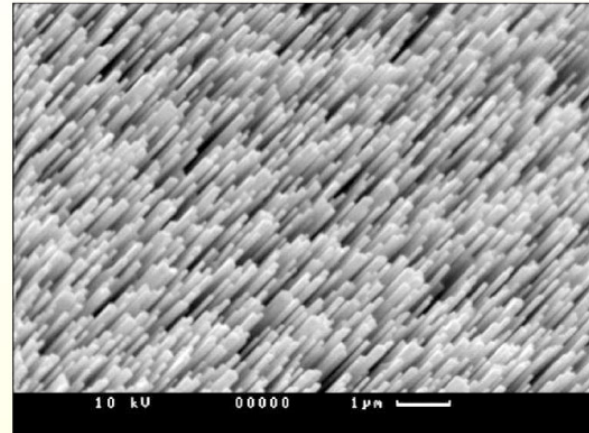
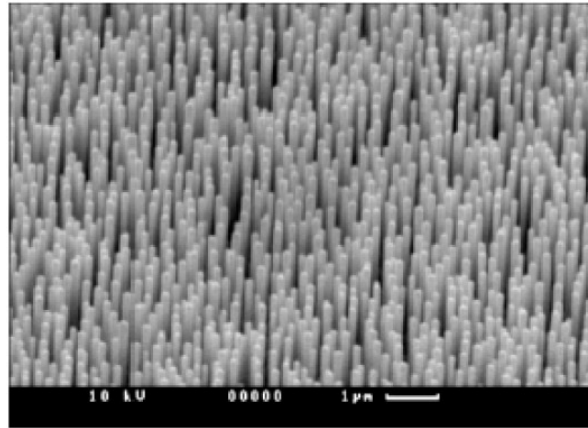


- investigation via microscopy,
- X-ray-tomography,
- geometrical properties in a statistical distribution

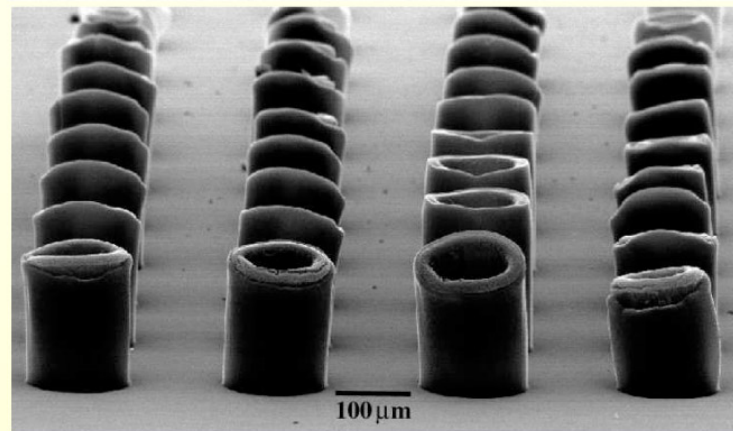
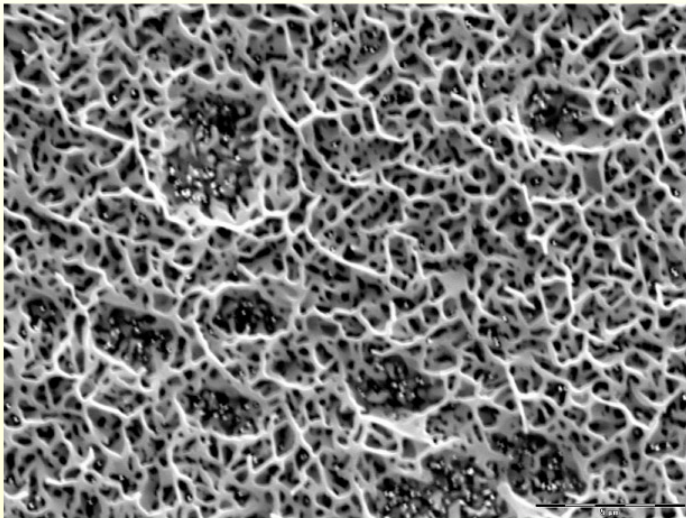
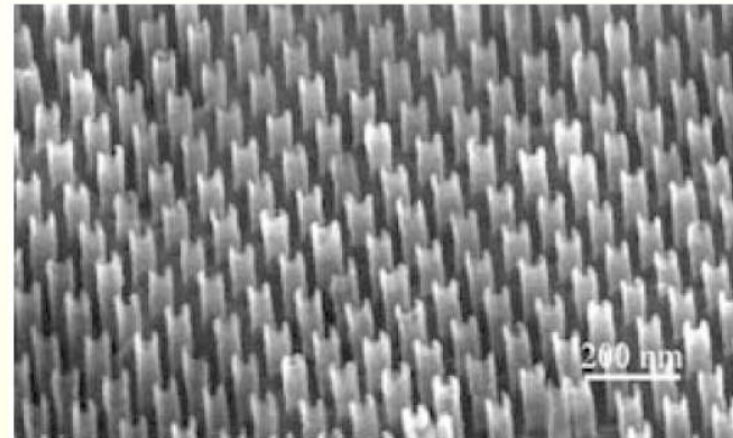
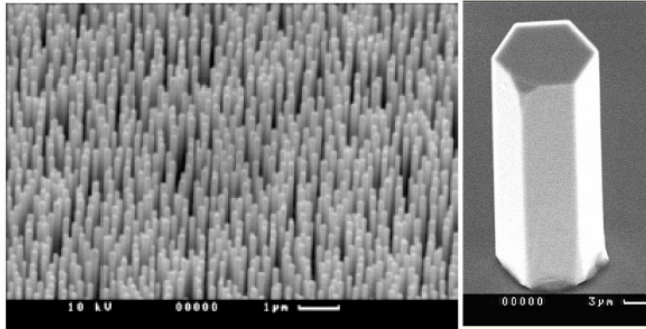
Three scales



Nanostructures

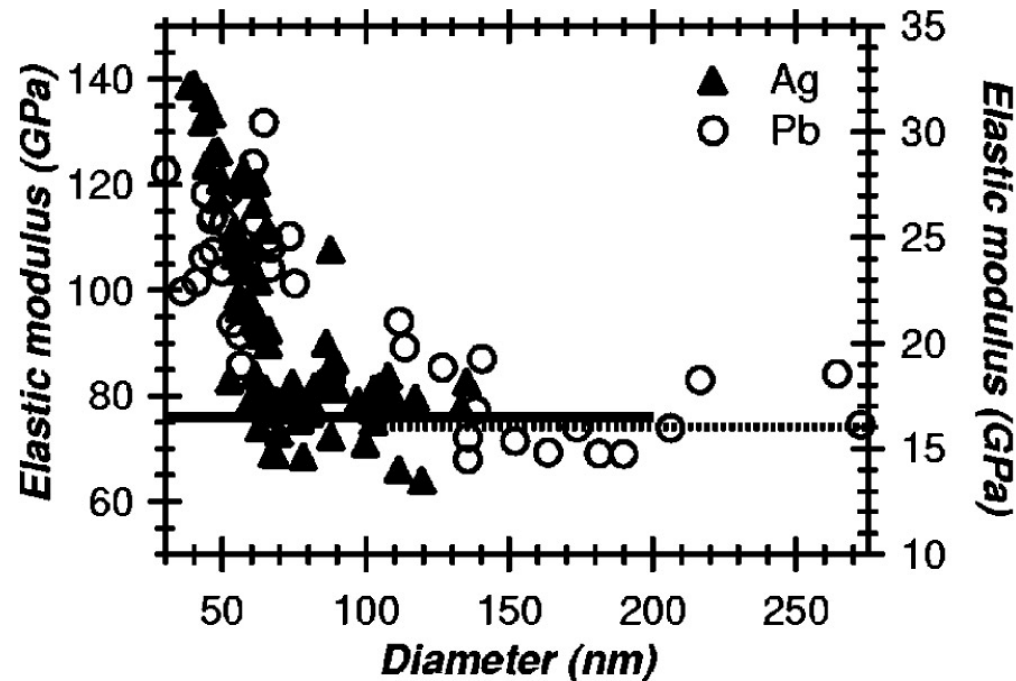


Nanostructures



Nano-scale Structure Behavior - Size Dependent³

Young's Modulus vs. Nanowire Diameter



solid line - elastic modulus of bulk silver

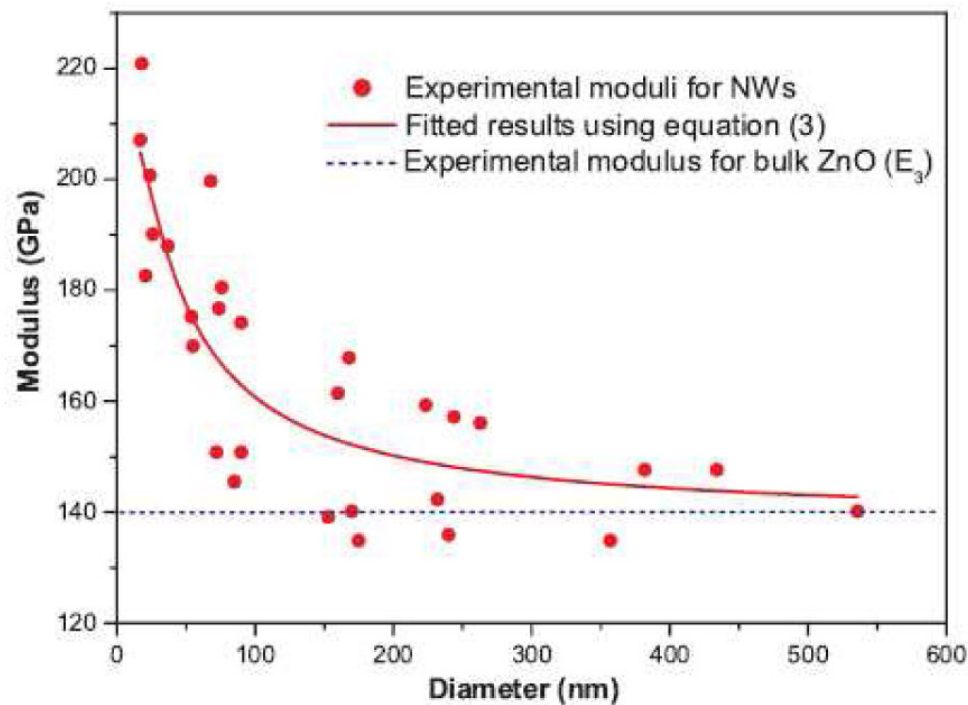
dotted line - elastic modulus of bulk lead

³S. Cuenot, Physical Review B **69**(2004) 165410 1-5

Experimental Observations

Surface stresses $\rightarrow$ size effect

Young's modulus, experimental data: eigenfrequencies of nanowires^a



^aChen et al. (2006)

Qualitative Microstructure Prediction

Aim of the modeling

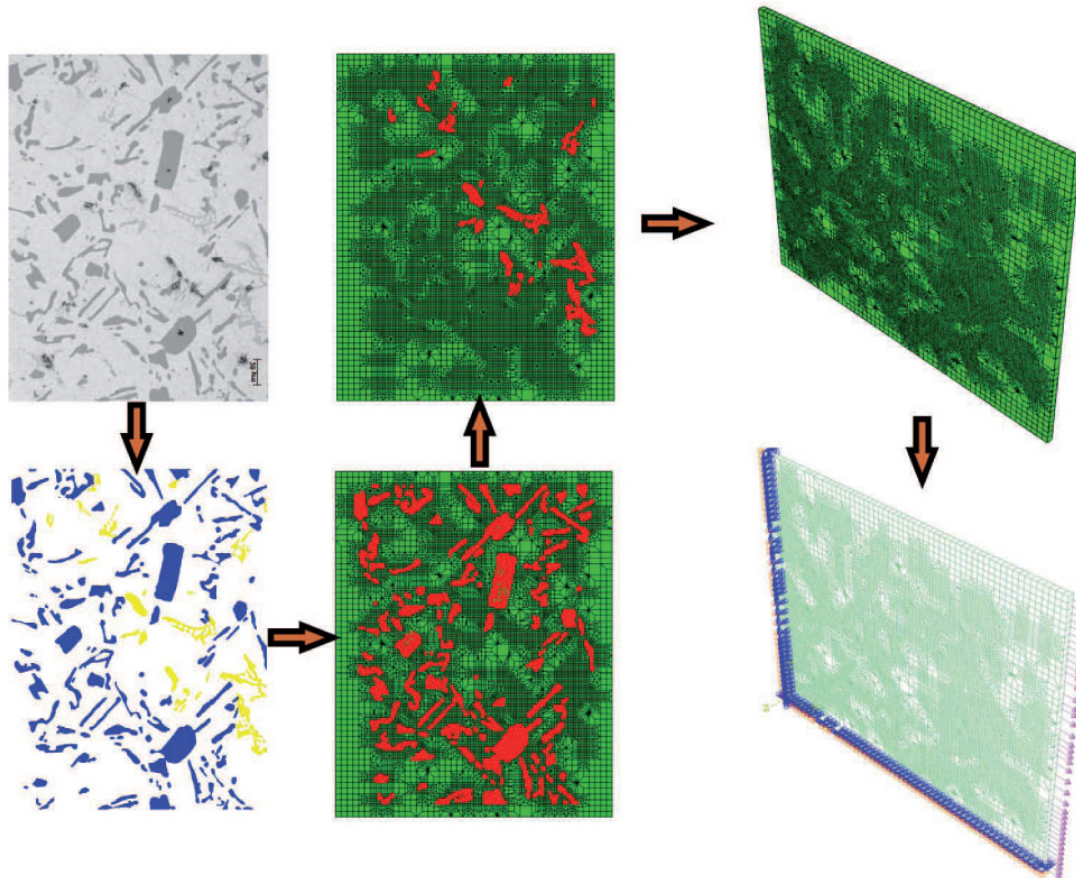
Investigation of the influence of the hard phases and the morphology of silicon on the behavior of the primary material in the original (RT6) and the improved (MT6) state with the help of micrographs.

Preliminary steps

- 1 Image analysis using the software *ImageJ*.
- 2 Image improvement and color coding of the phases using the software *CorelDraw*
- 3 Image input and FE meshing using of the software *OOF2*
- 4 Input in the FE software *ABAQUS*
- 5 Extrusion of the elements for three-dimensional model
- 6 Prescription of the boundary conditions

FE-Modeling

FE model from the RT6 microstructure



FE-Modeling

Material Parameters

		aluminium	intermetallic	silicon
surface fraction/volume fraction [%]	MT6	85,4	2,6	12
	RT6	85,7	2,3	12
thermal expansion coefficient	α [K ⁻¹]	$2,77 \cdot 10^{-5}$	$1,16 \cdot 10^{-5}$	$0,38 \cdot 10^{-5}$
elastic properties	E [GPa]	55	140	103
	ν	0,34	0,31	0,215
inelastic properties	$\dot{\epsilon}_{eq}^{cr} = a \sigma_{eq}^n$	$a = 3,3 \cdot 10^{-7} [\text{MPa}^{-n}/\text{h}]$ $n = 5$		—

Loading Parameters

- heating $T_1 = 20^\circ\text{C}$; $T_2 = 300^\circ\text{C}$
- monotonic creep $\sigma = 10 \text{ MPa}$
- cyclic creep $\sigma_{\max} = 10 \text{ MPa}$

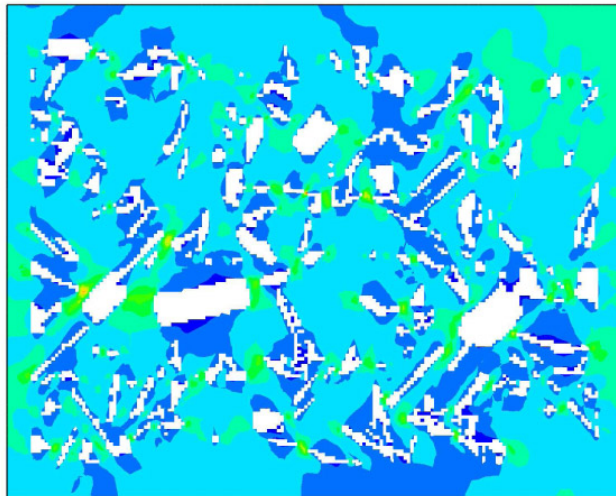
Modell Parameters

- size $0.295 \text{ mm} \times 0.236 \text{ mm}$
- hexaedric element C3D8
- number of elements
16034 – RT6, 19662 – MT6

Results of the FE-Modeling. Monotonic creep

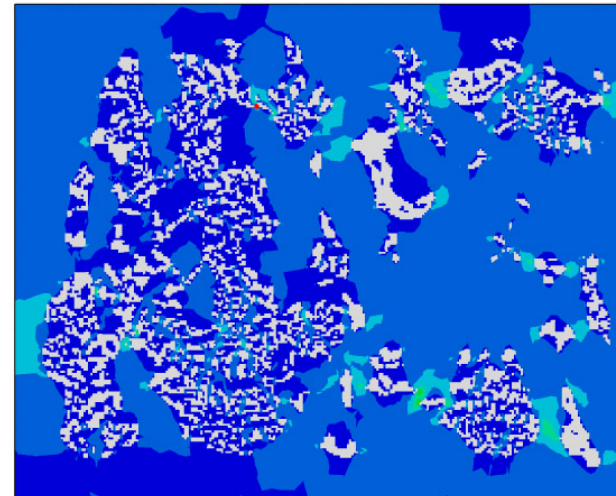
Distribution of the inelastic strain $\varepsilon_{xx}^{\text{in}}$ (SDV1; [m/m]) at static creep

SDV1
(Avg: 75%)
+5.7e-01
+5.1e-01
+4.4e-01
+3.8e-01
+3.2e-01
+2.6e-01
+1.9e-01
+1.3e-01
+6.7e-02
+4.4e-03
-5.8e-02



Material RT6

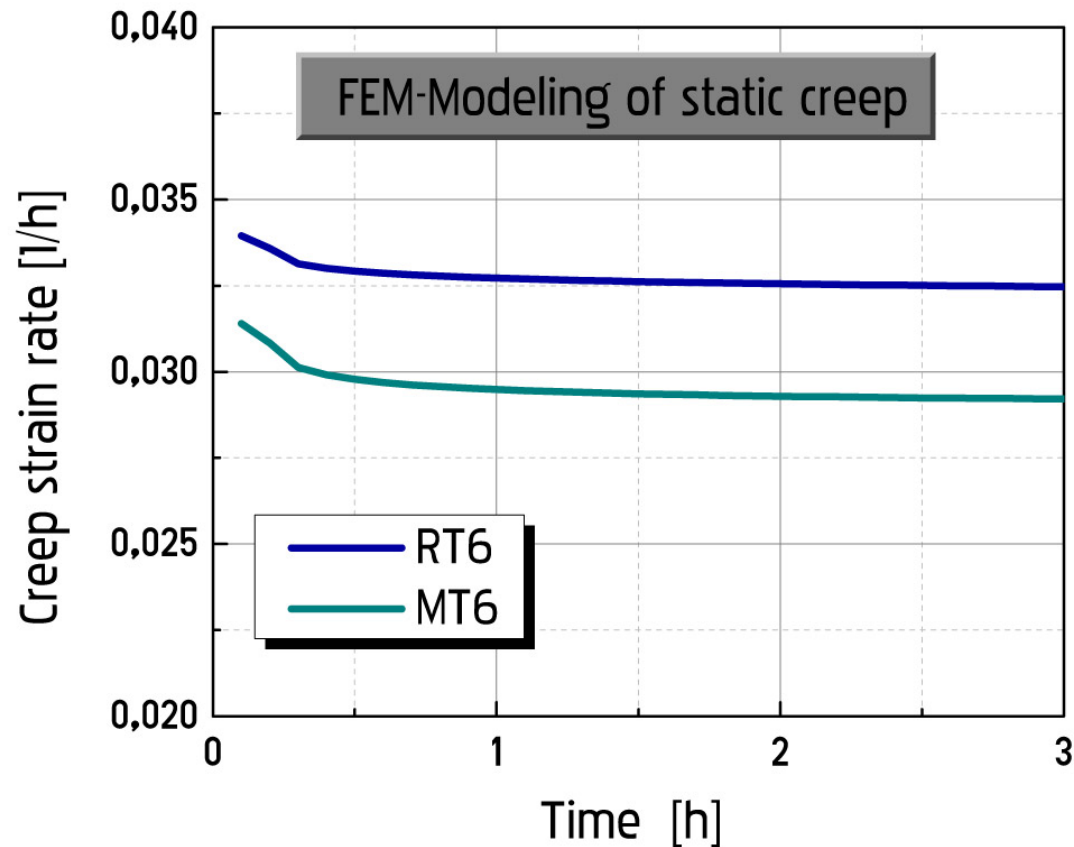
SDV1
(Avg: 75%)
+1.0e+00
+9.0e-01
+7.9e-01
+6.9e-01
+5.8e-01
+4.8e-01
+3.8e-01
+2.7e-01
+1.7e-01
+6.1e-02
-4.3e-02



Material MT6

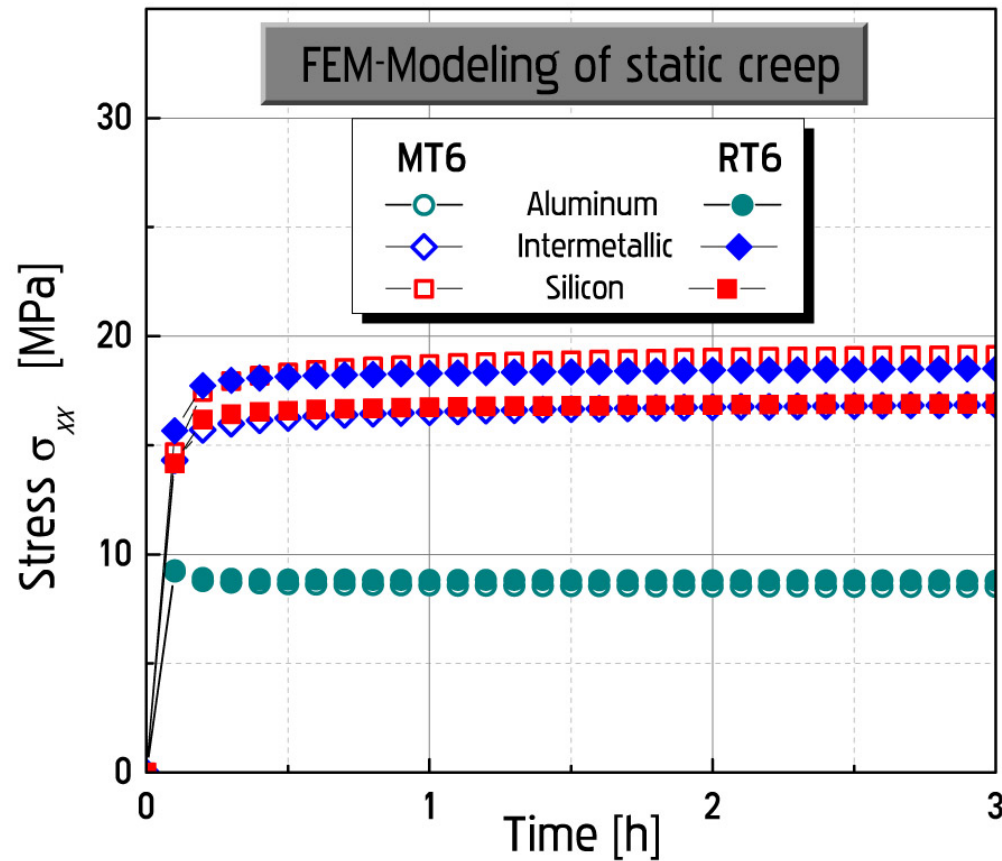
Results of the FE-Modeling. Monotonic creep

Creep Strain $\varepsilon_{xx}^{\text{in}}$ for Aluminium



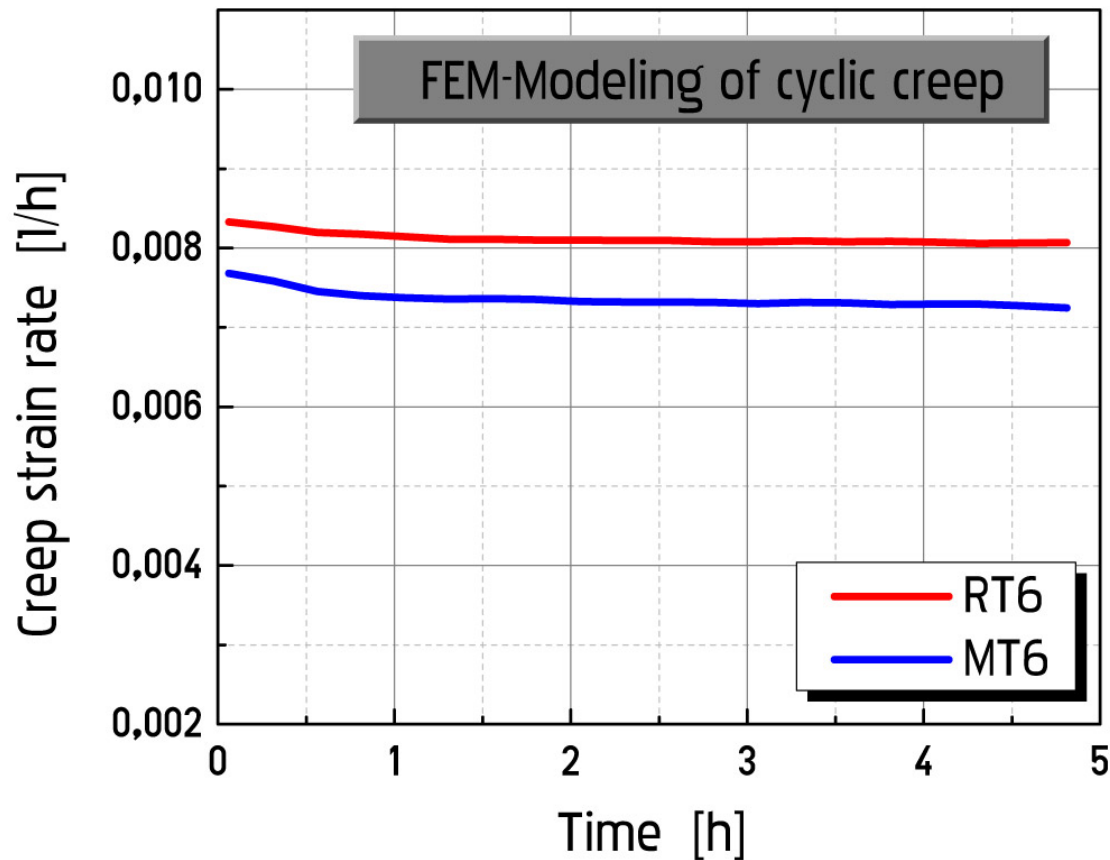
Results of the FE-Modeling. Monotonic creep

Averaged Stress Values σ_{xx} in the Constituents



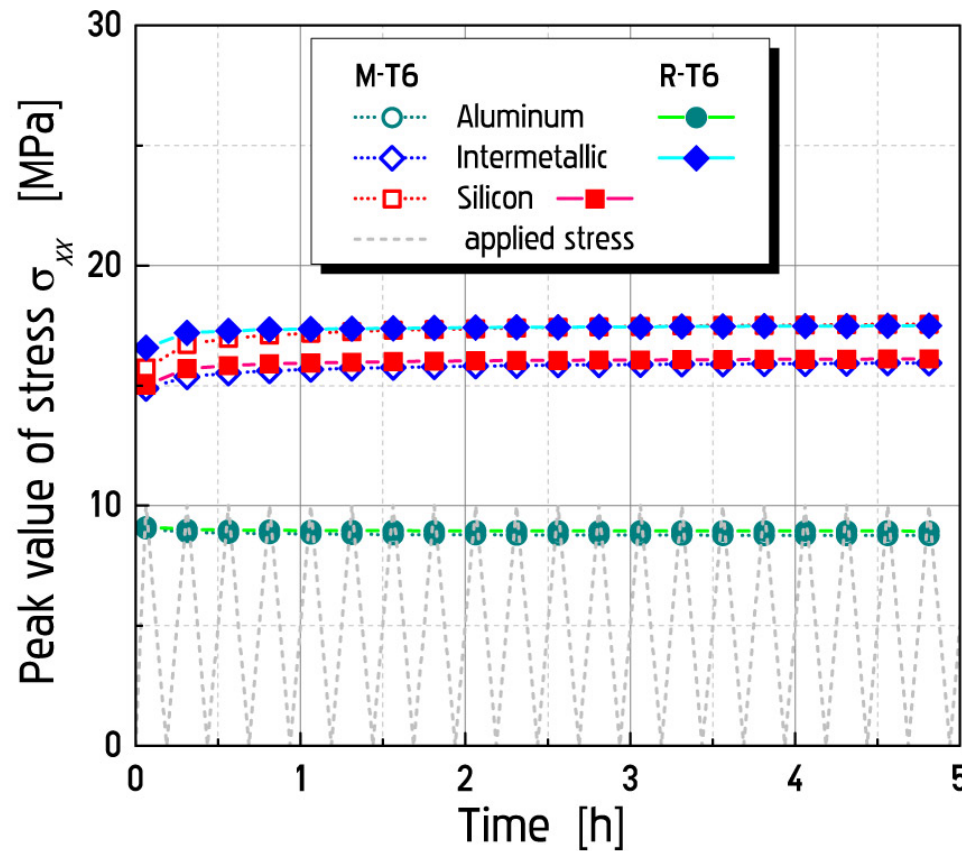
Results of the FE-Modeling. Cyclic creep

Maximum Creep Strain Rate $\dot{\epsilon}_{xx}^{in}$ of Aluminium



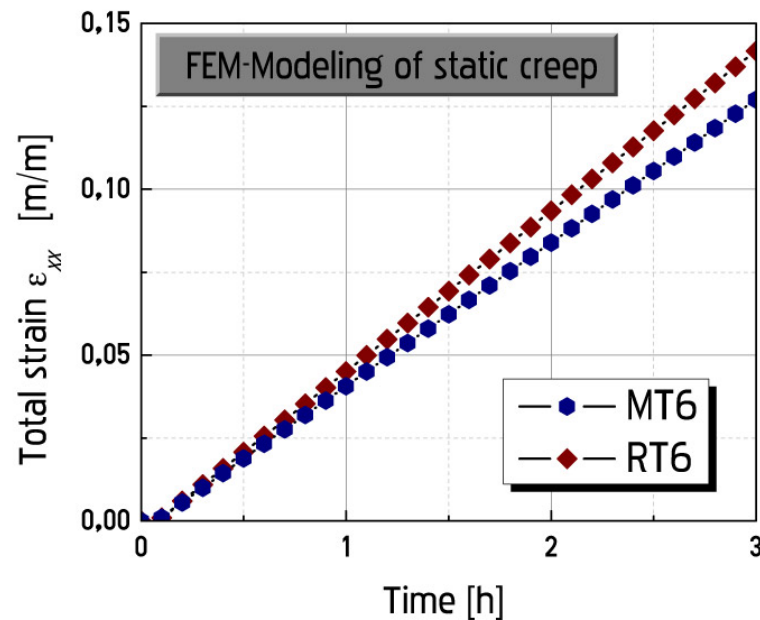
Results of the FE-Modeling. Cyclic creep

Averaged Stress Values σ_{xx} in the Constituents

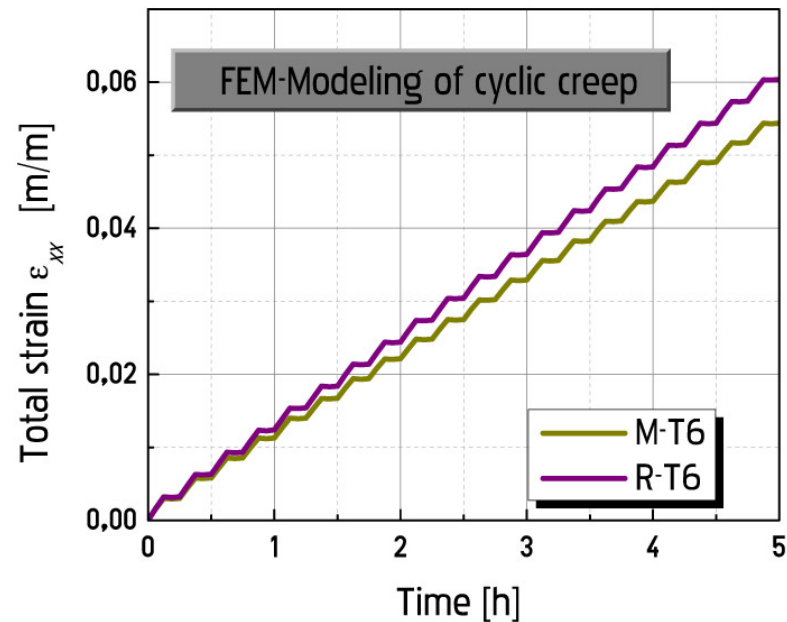


Results of the FE-Modeling. Static & Cyclic Creep

Averaged Values ε_{xx}

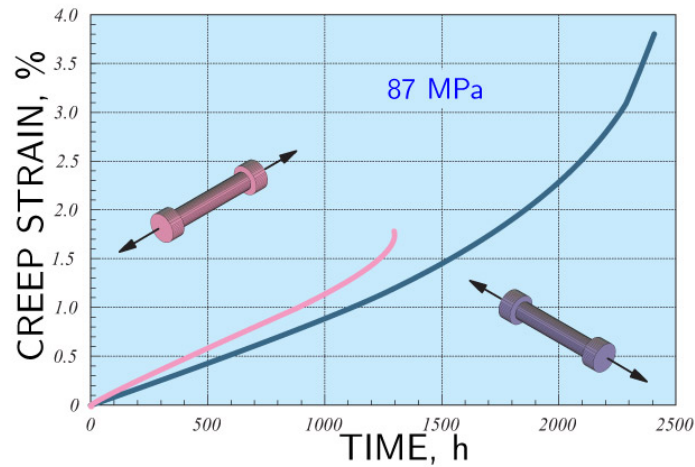
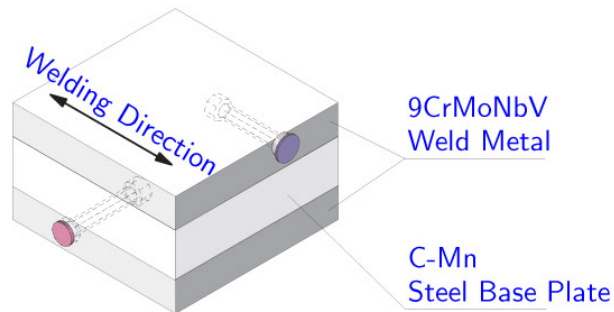
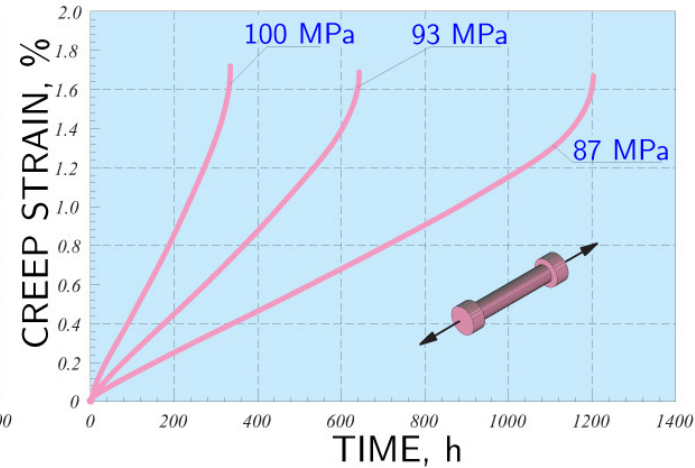
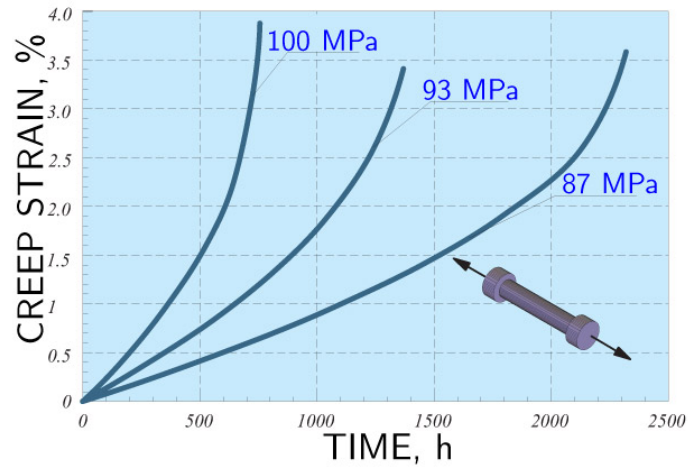


Averaged Values ε_{xx}



Creep Behavior of a Weld Metal

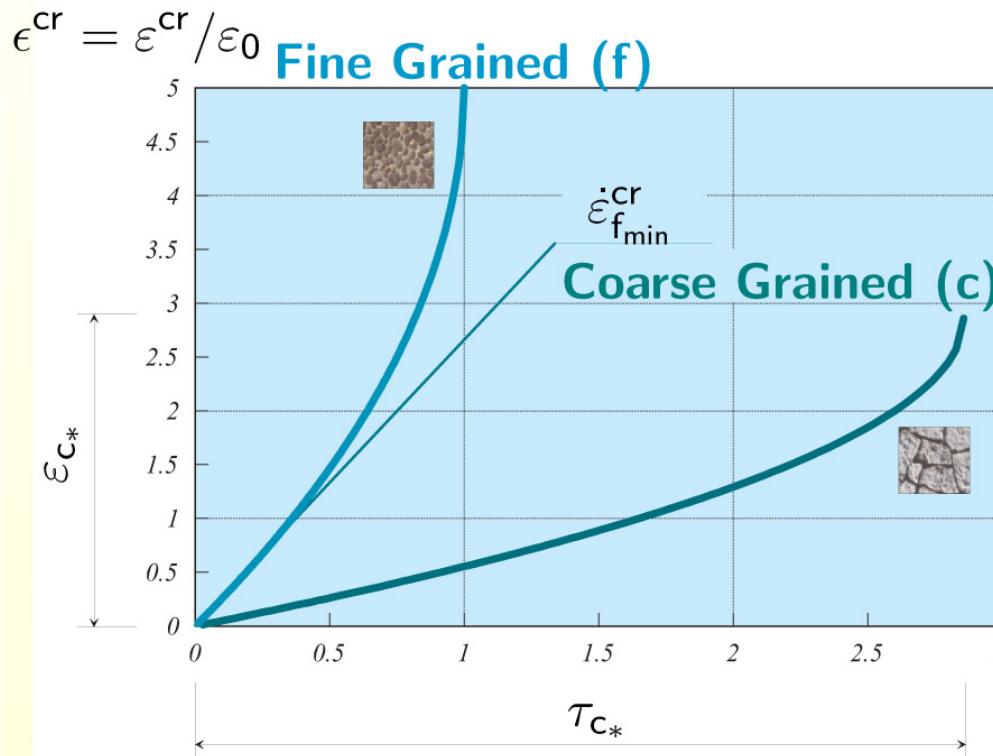
9CrMoNbV (P91) at 650° C (after Hyde et al., 2003)



Creep Behavior of Constituents (I)

Experimental data (grain size influence)

Kowalewski (1992), Wohlfahrt & Brinkmann (2001), Matsui et al. (2001), Wu et al. (2004)



$$\begin{cases} \dot{\epsilon}^{cr} = \frac{a\sigma^n}{(1-\omega)^n} \\ \dot{\omega} = \frac{b\sigma^k}{(1-\omega)^l} \end{cases}$$

$$\ln \dot{\epsilon}_{min}^{cr} = \ln a + n \ln \sigma$$

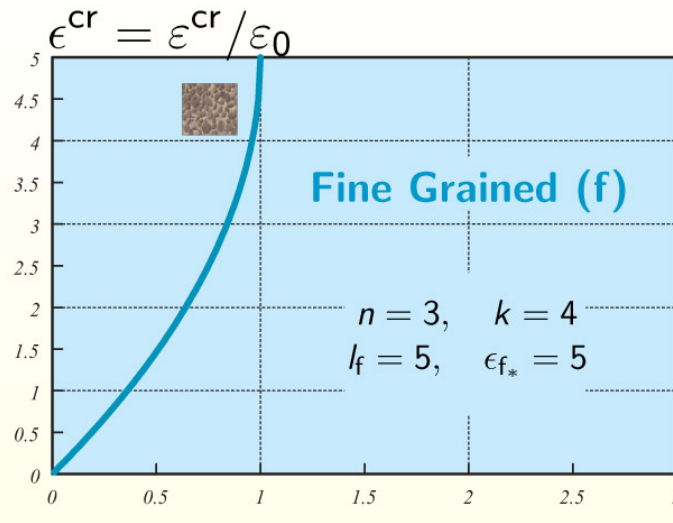
$$t_* = \frac{1}{(l+1)b\sigma^k}$$

$$\epsilon_*^{cr} = \frac{a\sigma^{n-k}}{b(l+1-n)}$$

Leckie & Hayhurst, (1977)

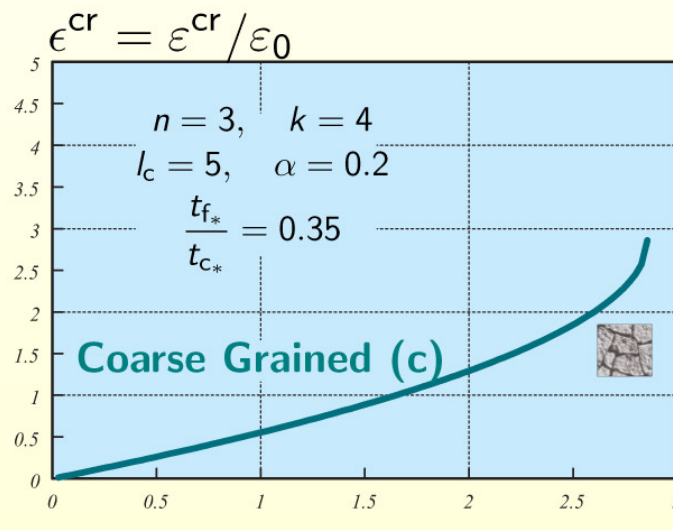
$$\tau = t / t_{f*}$$

Creep Behavior of Constituents (II)



$$\begin{cases} \frac{d}{d\tau} \epsilon_f^{cr} = \epsilon_{f*} \left(1 - \frac{n}{l_f + 1} \right) \left(\frac{\tilde{\sigma}}{1 - \omega_f} \right)^n \\ \frac{d}{d\tau} \omega_f = \frac{1}{l_f + 1} \frac{\tilde{\sigma}^k}{(1 - \omega_f)^{l_f}} \end{cases}$$

$$\tilde{\sigma} = \frac{\sigma}{\sigma_0}, \quad \epsilon_0 = \frac{\sigma_0}{E}$$



$$\begin{cases} \frac{d}{d\tau} \epsilon_c^{cr} = \alpha \epsilon_{f*} \left(1 - \frac{n}{l_f + 1} \right) \left(\frac{\tilde{\sigma}}{1 - \omega_c} \right)^n \\ \frac{d}{d\tau} \omega_c = \beta \frac{1}{l_f + 1} \frac{\tilde{\sigma}^k}{(1 - \omega_c)^{l_c}} \end{cases}$$

$$\alpha = \frac{\dot{\epsilon}_{cmin}^{cr}}{\dot{\epsilon}_{fmin}^{cr}}, \quad \beta = \frac{t_{f*}}{t_{c*}} \frac{l_f + 1}{l_c + 1}$$

Binary Structure (I)

- Assumptions:

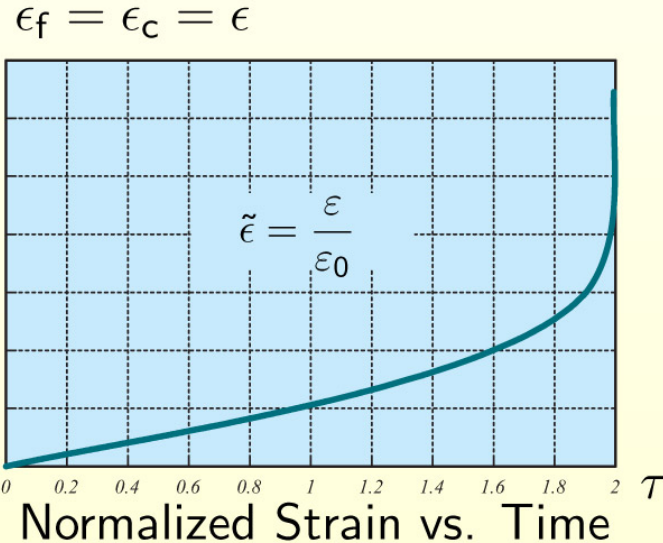
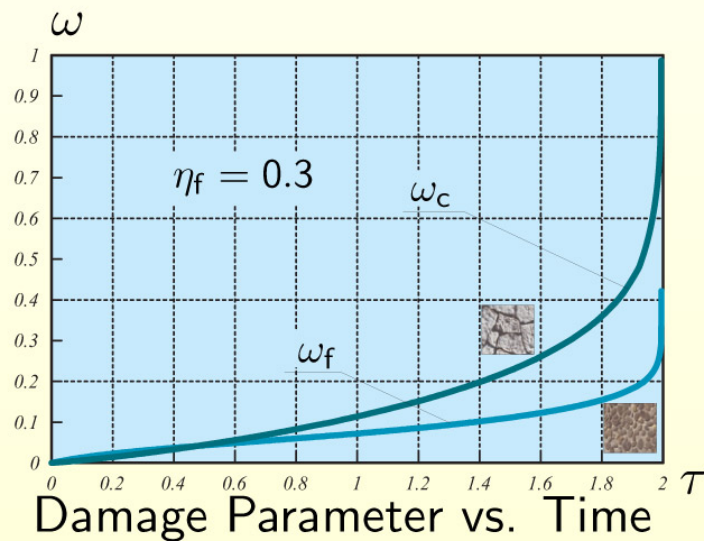
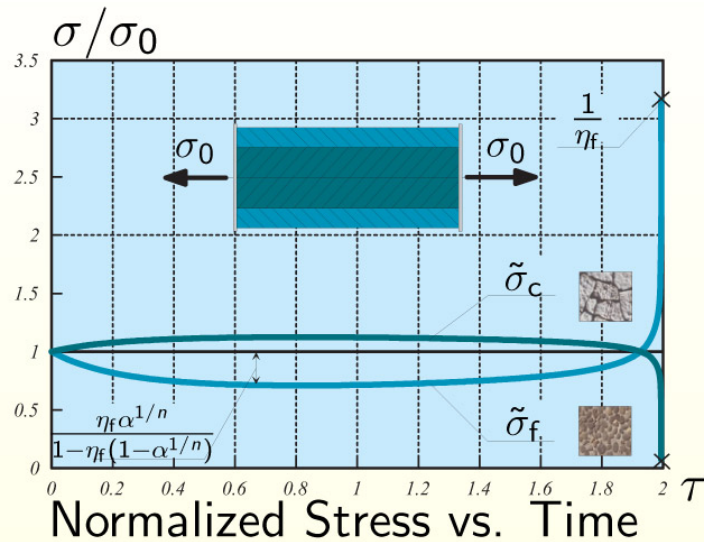
$$\varepsilon_f = \varepsilon_c \Rightarrow \dot{\varepsilon}_f = \dot{\varepsilon}_c,$$

$$\sigma_0 = \eta_f \sigma_f + (1 - \eta_f) \sigma_c = \text{const},$$

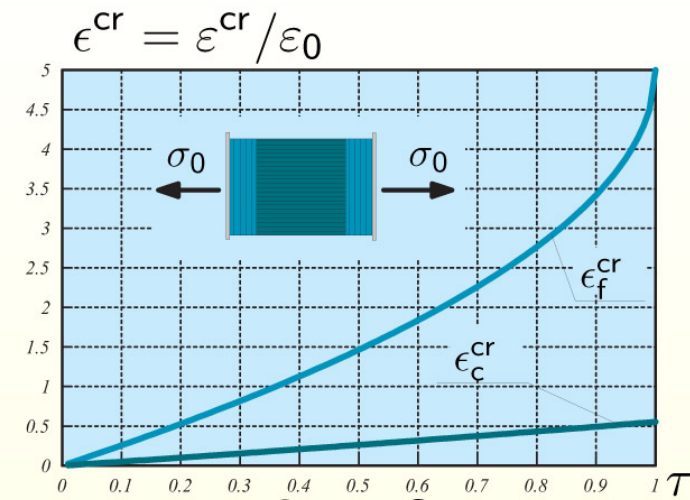
$$\sigma_f = E(1 - \omega_f)(\varepsilon - \varepsilon_f^{\text{cr}}),$$

$$\sigma_c = E(1 - \omega_c)(\varepsilon - \varepsilon_c^{\text{cr}}),$$

$$\begin{cases} \dot{\sigma}_f = g_1(\sigma_f, \omega_f, \omega_c) \\ \dot{\omega}_f = g_2(\sigma_f, \omega_f, \omega_c) \\ \dot{\omega}_c = g_3(\sigma_f, \omega_f, \omega_c) \end{cases}$$



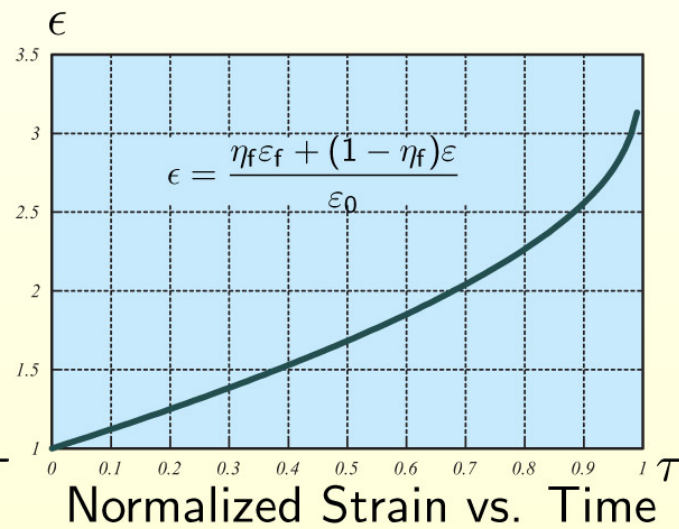
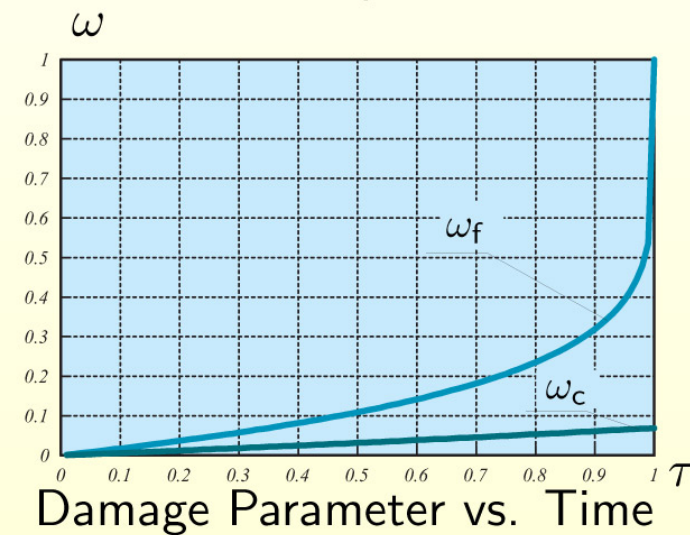
Binary Structure (II)



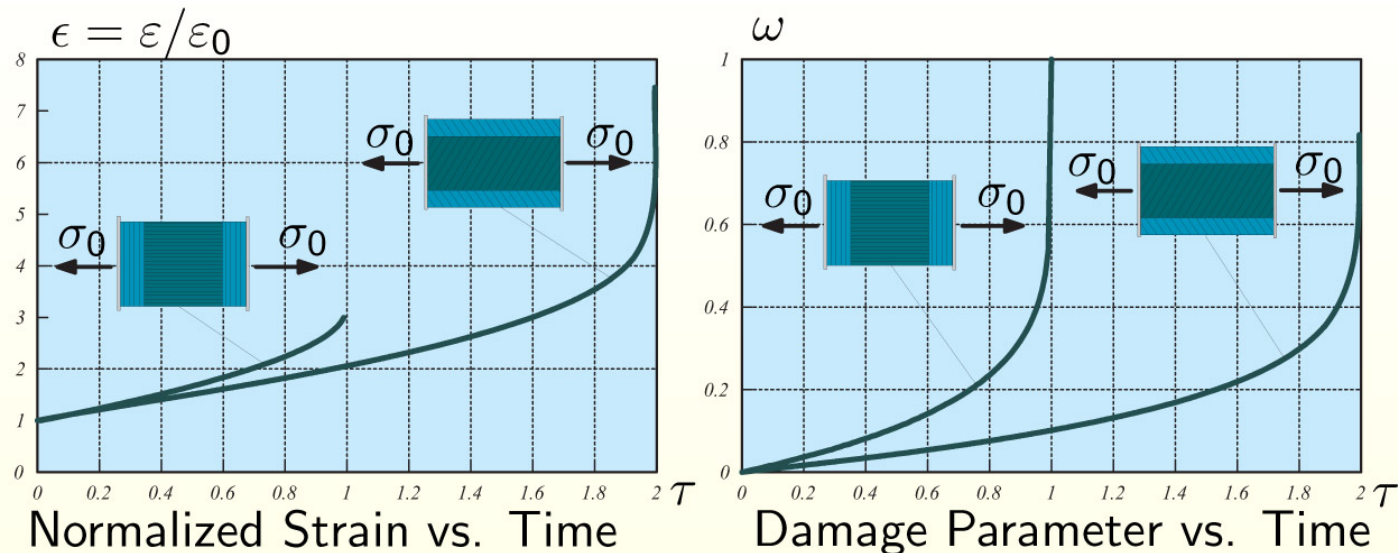
- Assumptions:

$$\sigma_0 = \sigma_f = \sigma_c = \text{const},$$

$$\epsilon = \eta_f \epsilon_f + (1 - \eta_f) \epsilon_c$$



Binary Structure (III)



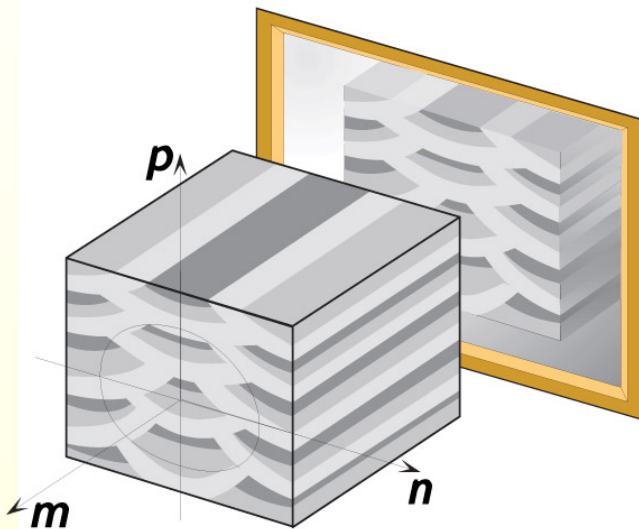
Microstructural observations of damage, Hyde et al. (2003)

- **Longitudinal specimen:** extensive voids and cracks along entire length of specimen, damage is confined to creep strong columnar regions
- **Transverse specimen:** voids and cracks are localized and are much less extensive away the fracture surface, the failure propagated through the fine-grained region

Symmetry Assumptions

Two Types of Symmetries

- Material Symmetries



- Reflection $Q_1 = E - 2m \otimes m$

- Rotation $Q_2(\pi p) = 2p \otimes p - E$

$\Downarrow$

- Reflection $Q_3 = Q_1 \cdot Q_2 = E - 2n \otimes n$

- Physical Symmetries

- Assumptions:

- a) **Transverse Isotropy:** $Q(\varphi m), \pm Q(\pi p), \quad -\pi < \varphi < \pi$

- b) **Orthotropic Symmetry:** $\pm Q(\pi n), \pm Q(\pi p)$

- Anisotropic Creep Models: Betten (2002)

Transversely-isotropic Creep

Restrictions

$$W(\mathbf{Q} \cdot \boldsymbol{\sigma} \cdot \mathbf{Q}^T) = W(\boldsymbol{\sigma})$$

$$\mathbf{Q} = \mathbf{m} \otimes \mathbf{m} + \cos\varphi(\mathbf{E} - \mathbf{m} \otimes \mathbf{m}) + \sin\varphi \mathbf{m} \times \mathbf{E}$$

$$-\pi < \varphi < \pi, \quad \mathbf{m} = \text{const}, \quad \mathbf{m} \cdot \mathbf{m} = 1$$

Invariants

$$W(\mathbf{Q} \cdot \boldsymbol{\sigma} \cdot \mathbf{Q}^T) = W(\boldsymbol{\sigma}) \quad \Rightarrow \quad (\mathbf{m} \times \boldsymbol{\sigma} - \boldsymbol{\sigma} \times \mathbf{m}) \cdot \frac{\partial W}{\partial \boldsymbol{\sigma}} = 0$$

$$\text{Characteristic system:} \quad \frac{d\boldsymbol{\sigma}}{ds} = (\mathbf{m} \times \boldsymbol{\sigma} - \boldsymbol{\sigma} \times \mathbf{m})$$

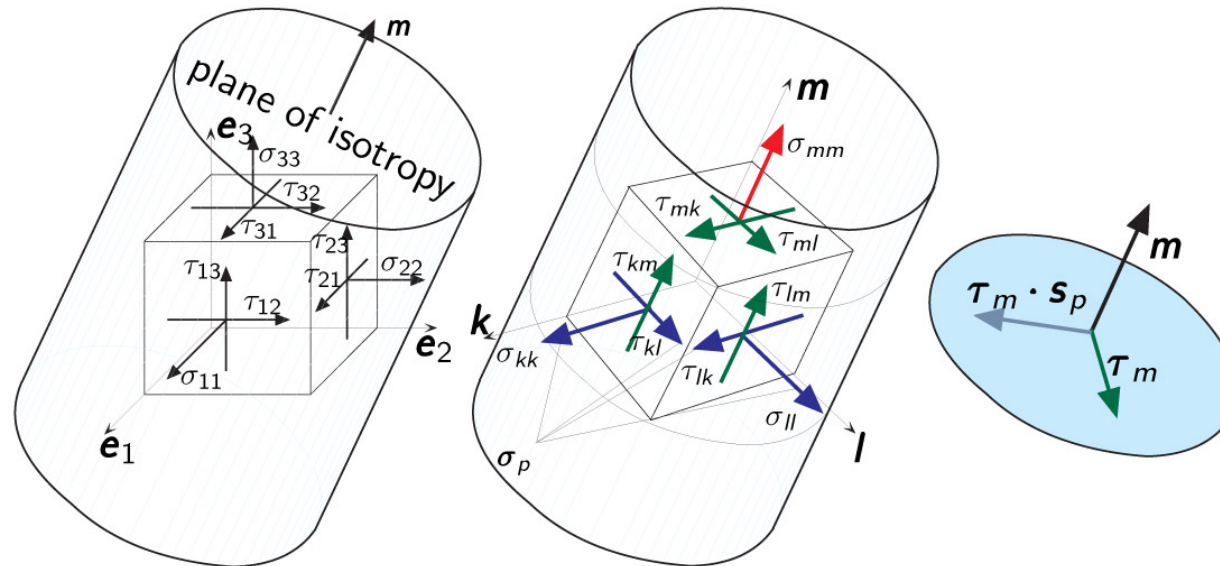
$$\text{Solutions:} \quad \boldsymbol{\sigma}^k(s) = \mathbf{Q}(s\mathbf{m}) \cdot \boldsymbol{\sigma}_0^k \cdot \mathbf{Q}^T(s\mathbf{m}), \quad k = 1, 2, 3$$

$$\text{Integrals:} \quad \text{tr}\boldsymbol{\sigma}, \quad \text{tr}\boldsymbol{\sigma}^2, \quad \text{tr}\boldsymbol{\sigma}^3, \\ \mathbf{m} \cdot \boldsymbol{\sigma} \cdot \mathbf{m}, \quad \mathbf{m} \cdot \boldsymbol{\sigma}^2 \cdot \mathbf{m}, \quad \mathbf{m} \cdot \boldsymbol{\sigma}^2 \cdot (\mathbf{m} \times \boldsymbol{\sigma} \cdot \mathbf{m})$$

Sets of transversely-isotropic invariants: Bruhns et al. (1999)

Transversely-isotropic Invariants

New Split of the Stress Tensor



$$\sigma = \mathbf{m} \cdot \sigma \cdot \mathbf{m} \mathbf{m} \otimes \mathbf{m} + \sigma_p + \tau_m \otimes \mathbf{m} + \mathbf{m} \otimes \tau_m,$$

$$\sigma_p = \mathbf{s}_p + \frac{1}{2} \text{tr} \sigma_p (\mathbf{E} - \mathbf{m} \otimes \mathbf{m}), \quad \text{tr} \mathbf{s}_p = 0,$$

$$l_{1m} = \mathbf{m} \cdot \sigma \cdot \mathbf{m}, \quad l_{2m} = \frac{1}{2} \text{tr} \sigma_p, \quad l_{3m} = \text{tr} \mathbf{s}_p^2, \quad l_{4m} = \tau_m \cdot \tau_m$$

$$l_{5m} = \tau_m \cdot \mathbf{s}_p \cdot \tau_m, \quad l_{6m} = \mathbf{m} \cdot (\tau_m \cdot \mathbf{s}_p \times \tau_m)$$

$$l_{6m}^2 = l_{3m} l_{4m} - l_{5m}^2$$

von Mises Type Creep Equation

- Equivalent Stress

$$\sigma_{\text{eq}}^2 = J_0^2 + 3\alpha_1 J_1 + 3\alpha_2 J_2, \quad \alpha_1 > 0, \quad \alpha_2 > 0$$

$$J_0 = \mathbf{m} \cdot \boldsymbol{\sigma} \cdot \mathbf{m} - \frac{1}{2} \text{tr} \boldsymbol{\sigma}_p = \frac{1}{2} (3\mathbf{m} \cdot \boldsymbol{\sigma} \cdot \mathbf{m} - \text{tr} \boldsymbol{\sigma})$$

$$\text{tr} \boldsymbol{\sigma}_p = \text{tr} \boldsymbol{\sigma} - \mathbf{m} \cdot \boldsymbol{\sigma} \cdot \mathbf{m}$$

$$2J_1 = \text{tr} \mathbf{s}_p^2 = \text{tr} \boldsymbol{\sigma}^2 - 2\mathbf{m} \cdot \boldsymbol{\sigma}^2 \cdot \mathbf{m} + (\mathbf{m} \cdot \boldsymbol{\sigma} \cdot \mathbf{m})^2 - \frac{1}{2} (\text{tr} \boldsymbol{\sigma}_p)^2$$

$$J_2 = \boldsymbol{\tau}_m \cdot \boldsymbol{\tau}_m = \mathbf{m} \cdot \boldsymbol{\sigma}^2 \cdot \mathbf{m} - (\mathbf{m} \cdot \boldsymbol{\sigma} \cdot \mathbf{m})^2$$

- Norton-Bailey-Odqvist Potential

$$W = \frac{a}{n+1} \sigma_{\text{eq}}^{n+1} \quad (\text{power law creep})$$

- Creep Equation

$$\mathbf{D}^{\text{cr}} = \frac{3}{2} a \sigma_{\text{eq}}^{n-1} \left[J_0 (\mathbf{m} \otimes \mathbf{m} - \frac{1}{3} \mathbf{E}) + \alpha_1 \mathbf{s}_p + \alpha_2 (\boldsymbol{\tau}_m \otimes \mathbf{m} + \mathbf{m} \otimes \boldsymbol{\tau}_m) \right]$$

- Classical Case: $\alpha_1 = \alpha_2 = 1$

Material Constants I

- Creep Equation

$$\mathbf{D}^{\text{cr}} = \frac{3}{2} a \sigma_{\text{eq}}^{n-1} \left[J_0 (\mathbf{m} \otimes \mathbf{m} - \frac{1}{3} \mathbf{E}) + \alpha_1 \mathbf{s}_p + \alpha_2 (\boldsymbol{\tau}_m \otimes \mathbf{m} + \mathbf{m} \otimes \boldsymbol{\tau}_m) \right]$$

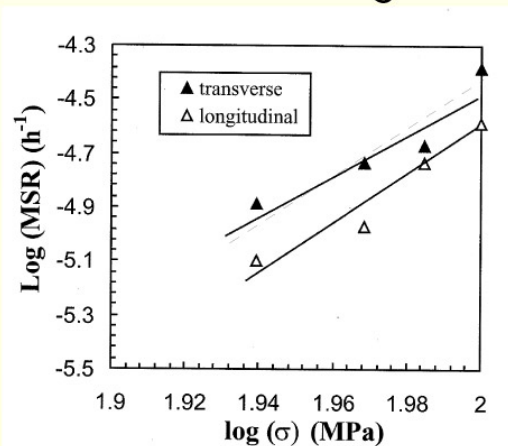
- Creep tests in longitudinal (welding) direction

$$\boldsymbol{\sigma} = \sigma_0 \mathbf{m} \otimes \mathbf{m} : \quad \dot{\epsilon}^{\text{cr}} = \mathbf{m} \cdot \mathbf{D}^{\text{cr}} \cdot \mathbf{m} = a \sigma_0^n, \quad \sigma_0 [87 - 100] \text{MPa}$$

- Creep tests in transverse direction

$$\boldsymbol{\sigma} = \sigma_0 \mathbf{p} \otimes \mathbf{p}, \quad \mathbf{p} \cdot \mathbf{m} = 0, \quad \dot{\epsilon}^{\text{cr}} = \mathbf{p} \cdot \mathbf{D}^{\text{cr}} \cdot \mathbf{p} = a \gamma^{n+1} \sigma_0^n,$$

$$\alpha_1 = \frac{1}{3} (4\gamma^2 - 1), \quad \sigma_0 [87 - 100] \text{MPa}$$



$$a = 1.15 \cdot 10^{-22} \text{MPa}^{-n}/\text{h}, \quad n = 8.64$$

$$\alpha_1 = 1.23$$

Minimum Creep Rate vs. Stress,
Hyde et al. (2003)

Material Constants I

- Creep Equation

$$\mathbf{D}^{\text{cr}} = \frac{3}{2} a \sigma_{\text{eq}}^{n-1} \left[J_0 (\mathbf{m} \otimes \mathbf{m} - \frac{1}{3} \mathbf{E}) + \alpha_1 \mathbf{s}_p + \alpha_2 (\boldsymbol{\tau}_m \otimes \mathbf{m} + \mathbf{m} \otimes \boldsymbol{\tau}_m) \right]$$

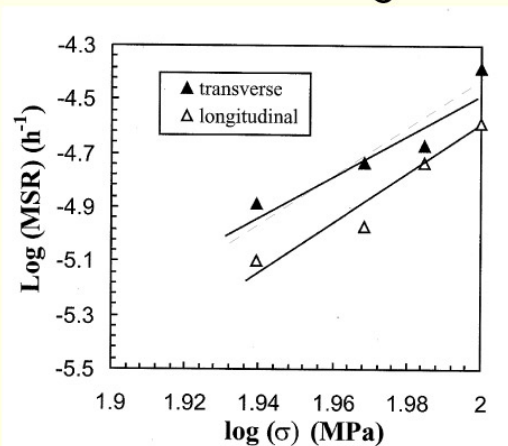
- Creep tests in longitudinal (welding) direction

$$\boldsymbol{\sigma} = \sigma_0 \mathbf{m} \otimes \mathbf{m} : \quad \dot{\epsilon}^{\text{cr}} = \mathbf{m} \cdot \mathbf{D}^{\text{cr}} \cdot \mathbf{m} = a \sigma_0^n, \quad \sigma_0 [87 - 100] \text{MPa}$$

- Creep tests in transverse direction

$$\boldsymbol{\sigma} = \sigma_0 \mathbf{p} \otimes \mathbf{p}, \quad \mathbf{p} \cdot \mathbf{m} = 0, \quad \dot{\epsilon}^{\text{cr}} = \mathbf{p} \cdot \mathbf{D}^{\text{cr}} \cdot \mathbf{p} = a \gamma^{n+1} \sigma_0^n,$$

$$\alpha_1 = \frac{1}{3} (4\gamma^2 - 1), \quad \sigma_0 [87 - 100] \text{MPa}$$



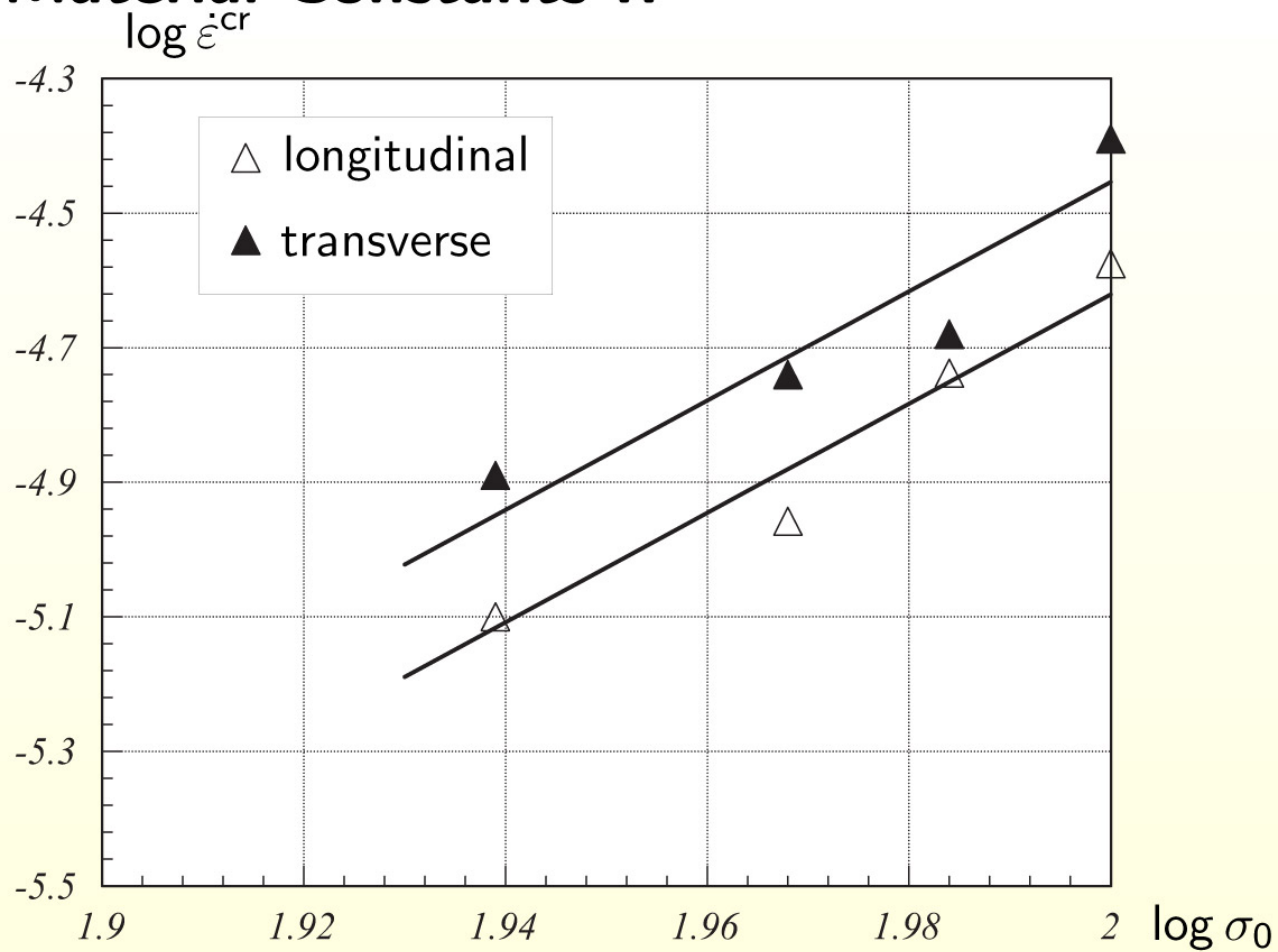
$$a = 1.15 \cdot 10^{-22} \text{MPa}^{-n}/\text{h}, \quad n = 8.64$$

$$\alpha_1 = 1.23$$

$$\alpha_2 = ??$$

Minimum Creep Rate vs. Stress,
Hyde et al. (2003)

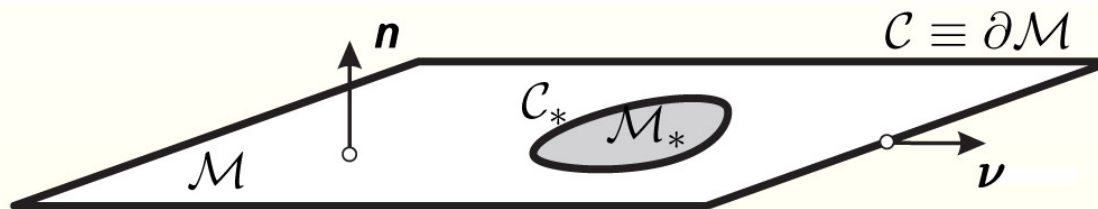
Material Constants II



$$a = 1.38 \cdot 10^{-21} \text{MPa}^{-n}/\text{h}, \quad n = 8.12, \quad \alpha_1 = 1.17$$

$$\alpha_2 = ??$$

Balance of Momentum and Moment of Momentum



Deformable plane surface

Integral Form

$$\mathcal{F}_s^* \equiv \int_{\mathcal{M}_*} \mathbf{q} \, dA + \int_{\mathcal{C}_*} \mathbf{t}_s \, ds = \mathbf{0},$$

$$\mathcal{M}_s^* \equiv \int_{\mathcal{M}_*} (\mathbf{x} \times \mathbf{q} + \mathbf{c}) \, dA + \int_{\mathcal{C}_*} (\mathbf{x} \times \mathbf{t}_s + \mathbf{m}_s) \, ds = \mathbf{0}$$

Constitutive Equations

Strain Energy Density - Isotropic Case

$$2W = \alpha_1 \text{tr}^2 \boldsymbol{\epsilon}_{\parallel} + \alpha_2 \text{tr} \boldsymbol{\epsilon}_{\parallel}^2 + \alpha_3 \text{tr} (\boldsymbol{\epsilon}_{\parallel} \cdot \boldsymbol{\epsilon}_{\parallel}^T) + \alpha_4 \mathbf{n} \cdot \boldsymbol{\epsilon}^T \cdot \boldsymbol{\epsilon} \cdot \mathbf{n} \\ + \beta_1 \text{tr}^2 \boldsymbol{\kappa}_{\parallel} + \beta_2 \text{tr} \boldsymbol{\kappa}_{\parallel}^2 + \beta_3 \text{tr} (\boldsymbol{\kappa}_{\parallel} \cdot \boldsymbol{\kappa}_{\parallel}^T) + \beta_4 \mathbf{n} \cdot \boldsymbol{\kappa}^T \cdot \boldsymbol{\kappa} \cdot \mathbf{n}$$

Here $\boldsymbol{\epsilon}_{\parallel} = \boldsymbol{\epsilon} \cdot \mathbf{A}$, $\boldsymbol{\kappa}_{\parallel} = \boldsymbol{\kappa} \cdot \mathbf{A}$, and α_i, β_i are the elastic constants, $i = 1, 2, 3, 4$.

$$\mathbf{T} \equiv \frac{\partial W}{\partial \boldsymbol{\epsilon}} = \alpha_1 \mathbf{A} \text{tr} \boldsymbol{\epsilon}_{\parallel} + \alpha_2 \boldsymbol{\epsilon}_{\parallel}^T + \alpha_3 \boldsymbol{\epsilon}_{\parallel} + \alpha_4 \boldsymbol{\epsilon} \cdot \mathbf{n} \otimes \mathbf{n},$$

$$\mathbf{M} \equiv \frac{\partial W}{\partial \boldsymbol{\kappa}} = \beta_1 \mathbf{A} \text{tr} \boldsymbol{\kappa}_{\parallel} + \beta_2 \boldsymbol{\kappa}_{\parallel}^T + \beta_3 \boldsymbol{\kappa}_{\parallel} + \beta_4 \boldsymbol{\kappa} \cdot \mathbf{n} \otimes \mathbf{n}$$

Elastic Moduli

Constraints

$$\begin{aligned} 2\alpha_1 + \alpha_2 + \alpha_3 &> 0, & \alpha_2 + \alpha_3 &> 0, & \alpha_3 - \alpha_2 &> 0, & \alpha_4 &> 0, \\ 2\beta_1 + \beta_2 + \beta_3 &> 0, & \beta_2 + \beta_3 &> 0, & \beta_3 - \beta_2 &> 0, & \beta_4 &> 0. \end{aligned}$$

Example Chróścielewski et al. (2004)

$$\begin{aligned} \alpha_1 &= C\nu, & \alpha_2 &= 0, & \alpha_3 &= C(1 - \nu), & \alpha_4 &= \alpha_s C(1 - \nu), \\ \beta_1 &= D\nu, & \beta_2 &= 0, & \beta_3 &= D(1 - \nu), & \beta_4 &= \alpha_t D(1 - \nu), \\ C &= \frac{Eh}{1 - \nu^2}, & D &= \frac{Eh^3}{12(1 - \nu^2)}, \end{aligned}$$

where α_s and α_t are dimensionless coefficients.

Further Steps

- Three-dimensional Micropolar Elasticity
- Through-the-thickness integration procedure
- Micropolar Plates Constitutive Equations

Micropolar Constants for Two Materials⁴

		Foam, PU	Foam, PS
Shear modulus, MPa	$G = \frac{2\mu+\kappa}{2}$	1.1	104
Poisson's ratio,	$\nu = \frac{\lambda}{2\lambda+2\mu+\kappa}$	0.07	0.4
Characteristic length (torsion), mm	$l_t = \sqrt{\frac{\beta+\gamma}{2\mu+\kappa}}$	3.8	0.62
Characteristic length (bending), mm	$l_b = \sqrt{\frac{\gamma}{2(2\mu+\kappa)}}$	5.0	0.33
Coupling number,	$N^2 = \frac{\kappa}{2\mu+\kappa}$	0.09	0.04
Polar ratio,	$\Psi = \frac{\beta+\gamma}{\alpha+\beta+\gamma}$	1.5	1.5

- PS - low-density polystyrene closed-cell foam
- PU - high-density rigid polyurethane closed-cell foam

⁴see Lakes (1986, 1995)

Effective Stiffness for PU Foam

Elastic constants		Foam, PU	Foam, PU *
$\alpha_1, \text{N/m } 10^6$	$\lambda^* h$	$0.165h$	$0.165h$
$\alpha_2, \text{N/m } 10^6$	μh	$1.001h$	$1.1h$
$\alpha_3, \text{N/m } 10^6$	$(\mu + \kappa)h$	$1.199h$	$1.1h$
$\alpha_4, \text{N/m } 10^6$	$(\mu + \kappa)h$	$1.199h$	$1.1h$
$\beta_1, \text{N}\cdot\text{m } 10^6$	$\alpha h - \mu \frac{h^3}{12}$	$-2.6 \cdot 10^{-6}h - 0.083h^3$	$-0.092h^3$
$\beta_2, \text{N}\cdot\text{m } 10^6$	$\beta h - \lambda^* \frac{h^3}{12}$	$-10^{-4}h - 0.014h^3$	$-0.014h^3$
$\beta_3, \text{N}\cdot\text{m } 10^6$	$\gamma h + (2\mu + \kappa + \lambda^*) \frac{h^3}{12}$	$1.1 \cdot 10^{-4}h + 0.197h^3$	$0.197h^3$
$\beta_4, \text{N}\cdot\text{m } 10^6$	γh	$1.1 \cdot 10^{-4}h$	0

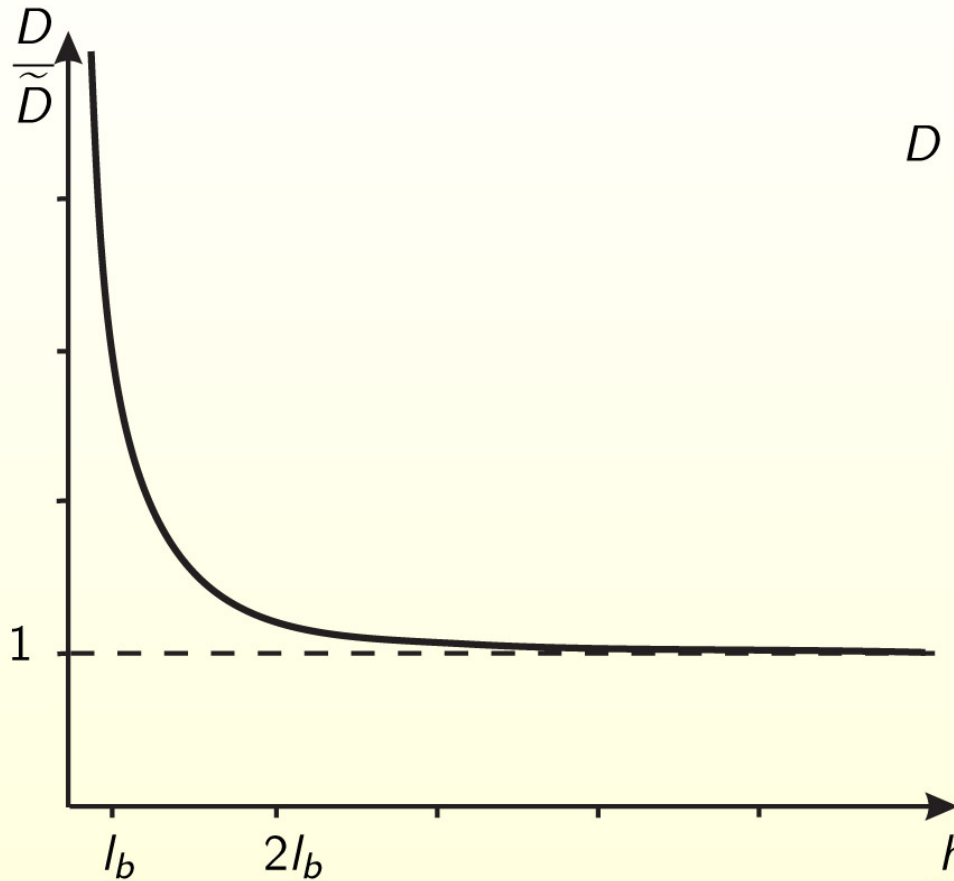
(...)* - micropolar properties are neglected ($\kappa = \alpha = \beta = \gamma = 0$)

Effective Stiffness for PS Foam

Elastic constants		Foam, PS	Foam, PS *
$\alpha_1, \text{N/m } 10^6$	$\lambda^* h$	$138.67h$	$138.67h$
$\alpha_2, \text{N/m } 10^6$	μh	$99.84h$	$104h$
$\alpha_3, \text{N/m } 10^6$	$(\mu + \kappa)h$	$108.16h$	$104h$
$\alpha_4, \text{N/m } 10^6$	$(\mu + \kappa)h$	$108.16h$	$104h$
$\beta_1, \text{N}\cdot\text{m } 10^6$	$\alpha h - \mu \frac{h^3}{12}$	$-6.7 \cdot 10^{-6}h - 8.3h^3$	$-8.67h^3$
$\beta_2, \text{N}\cdot\text{m } 10^6$	$\beta h - \lambda^* \frac{h^3}{12}$	$-2.5 \cdot 10^{-5}h + 11.6h^3$	$-11.5h^3$
$\beta_3, \text{N}\cdot\text{m } 10^6$	$\gamma h + (2\mu + \kappa + \lambda^*) \frac{h^3}{12}$	$4.5 \cdot 10^{-5}h + 28.9h^3$	$28.8h^3$
$\beta_4, \text{N}\cdot\text{m } 10^6$	γh	$4.5 \cdot 10^{-5}h$	0

(...)* - micropolar properties are neglected ($\kappa = \alpha = \beta = \gamma = 0$)

Dimensionless Bending Stiffness



$$D = \frac{Gh^3}{12(1-\nu)} \left[1 + 2\frac{l_b^2}{h^2} \right]$$

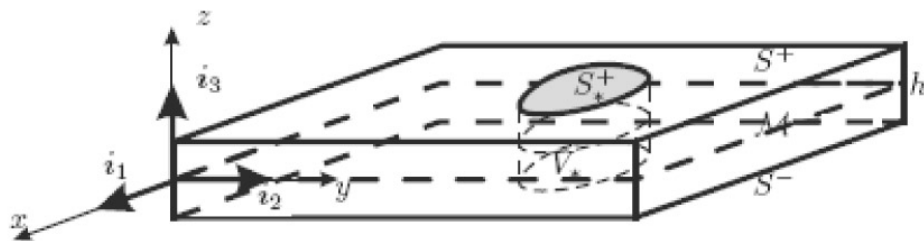
Dimensionless bending stiffness $D/\tilde{D}$ vs. thickness h/l_b

Theory of Capillarity

Surface Tension in the Hydrodynamics



Theory of Elasticity with Surface Stresses



Equilibrium equations

$$\nabla \cdot \boldsymbol{\sigma} + \rho \mathbf{f} = 0, \quad \mathbf{n} \cdot \boldsymbol{\sigma}|_{z=\pm \frac{h}{2}} = \mathbf{t}_{\pm}.$$

Surface loads

$$\mathbf{t}_{\pm} = \mathbf{t}_{\pm}^0 + \mathbf{t}_{\pm}^S, \quad \mathbf{t}_{\pm}^S = \nabla_S \cdot \boldsymbol{\tau}_{\pm},$$

Constitutive equations

$$\boldsymbol{\sigma} = 2\mu \boldsymbol{\epsilon} + \lambda \text{tr} \boldsymbol{\epsilon}, \quad \boldsymbol{\tau}_{\pm} = 2\mu_{\pm}^S \boldsymbol{\epsilon} + \lambda_{\pm}^S \mathbf{A} \text{tr} \boldsymbol{\epsilon},$$

Geometrical equations

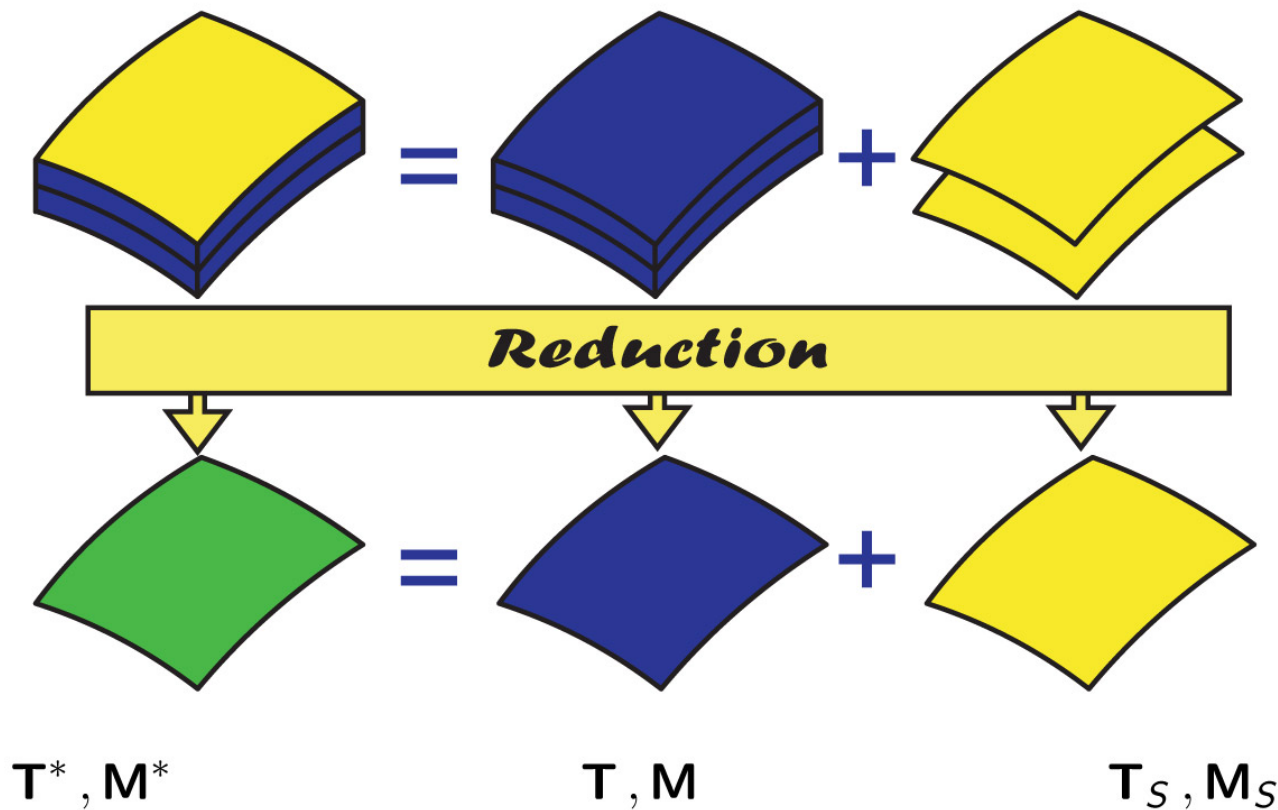
$$\boldsymbol{\epsilon} = \frac{1}{2} \left(\nabla \mathbf{u} + (\nabla \mathbf{u})^T \right), \quad \boldsymbol{\epsilon}_{\pm} = \frac{1}{2} \left(\nabla_S \mathbf{v}_{\pm} + (\nabla_S \mathbf{v}_{\pm})^T \right),$$

$\mathbf{I}$ - unit tensor, $\mathbf{A} \equiv \mathbf{I} - \mathbf{i}_3 \otimes \mathbf{i}_3$, $\mathbf{v}_{\pm} = \mathbf{v}_{\pm}(x, y) \equiv \mathbf{A} \cdot \mathbf{u}(x, y, \pm h/2)$

3D to 2D Reduction

Superposition of two problems

$$\mathbf{T}^* = \mathbf{T} + \mathbf{T}_S, \quad \mathbf{M}^* = \mathbf{M} + \mathbf{M}_S,$$



Transition to 2D Equations⁵

- Averaged three-dimensional equation

$$\nabla_S \cdot \langle \boldsymbol{\sigma} \rangle + \mathbf{t}_+ + \mathbf{t}_- + \langle \rho \mathbf{f} \rangle = 0,$$

$$-\nabla_S \cdot \langle z \boldsymbol{\sigma} \times \mathbf{i}_3 \rangle - \langle \mathbf{i}_3 \times \boldsymbol{\sigma} \cdot \mathbf{i}_3 \rangle + \frac{h}{2} \mathbf{i}_3 \times (\mathbf{t}_+ - \mathbf{t}_-) + \langle \rho z \mathbf{i}_3 \times \mathbf{f} \rangle = 0,$$

$$\langle (\dots) \rangle = \int_{-h/2}^{h/2} (\dots) dz$$

- Resultant stresses and moment stress tensors

$$\mathbf{T} = \mathbf{A} \cdot \langle \boldsymbol{\sigma} \rangle, \quad \mathbf{M} = -\mathbf{A} \cdot \langle z \boldsymbol{\sigma} \times \mathbf{i}_3 \rangle,$$

$$\nabla_S \cdot \mathbf{T} + \nabla_S \cdot \boldsymbol{\tau}_+ + \nabla_S \cdot \boldsymbol{\tau}_- + \mathbf{q} = 0,$$

$$\nabla_S \cdot \mathbf{M} + \mathbf{T}_\times + \frac{h}{2} \mathbf{i}_3 \times (\nabla_S \cdot \boldsymbol{\tau}_+ - \nabla_S \cdot \boldsymbol{\tau}_-) + \mathbf{m} = 0,$$

$$\mathbf{q} = \mathbf{t}_+^0 + \mathbf{t}_-^0 + \langle \rho \mathbf{f} \rangle, \quad \mathbf{m} = \mathbf{i}_3 \times (\mathbf{t}_+^0 - \mathbf{t}_-^0) h/2 + \langle \rho z \mathbf{i}_3 \times \mathbf{f} \rangle$$

⁵Eremeyev, Altenbach & Morozov (2009)

Effective Stress and Moment Stress Tensors

- Resultant tensors

$$\mathbf{T}^* = \mathbf{T} + \boldsymbol{\tau}_+ + \boldsymbol{\tau}_-, \quad \mathbf{M}^* = \mathbf{M} - \frac{h}{2}(\boldsymbol{\tau}_+ - \boldsymbol{\tau}_-) \times \mathbf{i}_3.$$

- Equilibrium equations

$$\nabla_S \cdot \mathbf{T}^* + \mathbf{q} = 0, \quad \nabla_S \cdot \mathbf{M}^* + \mathbf{T}_\times^* + \mathbf{m} = 0.$$

Constitutive Equations (I)

- Symmetric structure, i.e. $\lambda_+^S = \lambda_-^S = \lambda^S$, $\mu_+^S = \mu_-^S = \mu^S$. Then

$$\boldsymbol{\tau}_+ + \boldsymbol{\tau}_- = 2\lambda^S \mathbf{A} \text{tr} \boldsymbol{\epsilon} + 4\mu^S \boldsymbol{\epsilon}, \quad \boldsymbol{\tau}_+ - \boldsymbol{\tau}_- = -h \left[\lambda^S \mathbf{A} \text{tr} \boldsymbol{\kappa} + 2\mu^S \boldsymbol{\kappa} \right]$$

- Reissner-type constitutive equations

$$\begin{aligned} \mathbf{T} \cdot \mathbf{A} &= C [(1 - \nu) \boldsymbol{\epsilon} + \nu \mathbf{A} \text{tr} \boldsymbol{\epsilon}], \quad \mathbf{M} = D [(1 - \nu) \boldsymbol{\kappa} + \nu \mathbf{A} \text{tr} \boldsymbol{\kappa}] \times \mathbf{i}_3, \\ \mathbf{T} \cdot \mathbf{i}_3 &= \Gamma \boldsymbol{\gamma}, \quad C = \frac{Eh}{1 - \nu^2}, \quad D = \frac{Eh^3}{12(1 - \nu^2)}, \quad \Gamma = k\mu h, \end{aligned}$$

C and D - tangential and bending stiffness of a plate,

Γ - transverse shear stiffness,

$$E = 2\mu(1 + \nu), \quad \nu = \frac{\lambda}{2(\lambda + \mu)},$$

k the analogous to the shear correction factor, $\boldsymbol{\gamma} = \nabla_S w - \boldsymbol{\vartheta}$,

$w = \mathbf{w} \cdot \mathbf{i}_3$ the normal displacement

Constitutive Equations (II)

$$\mathbf{T}^* = C_1 \boldsymbol{\epsilon} + C_2 \mathbf{A} \text{tr} \boldsymbol{\epsilon} + \Gamma \boldsymbol{\gamma} \otimes \mathbf{i}_3, \quad \mathbf{M}^* = [D_1 \boldsymbol{\kappa} + D_2 \mathbf{A} \text{tr} \boldsymbol{\kappa}] \times \mathbf{i}_3$$

with

$$C_1 = C(1 - \nu) + 4\mu^S, \quad C_2 = C\nu + 2\lambda^S,$$
$$D_1 = D(1 - \nu) + h^2\mu^S, \quad D_2 = D\nu + h^2\lambda^S/2$$

Here: $\mathbf{T}^* \cdot \mathbf{i}_3 = \mathbf{T} \cdot \mathbf{i}_3$.

Conclusion:

Five independent elastic constants: $C_1, C_2, D_1, D_2, \Gamma$

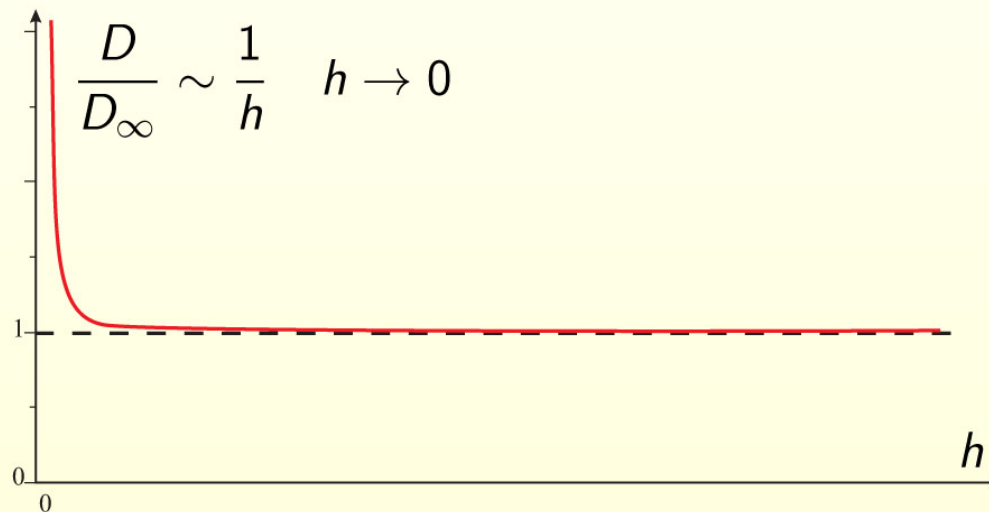
Reissner's theory: four constants only C, D, ν, Γ

Bending Stiffness

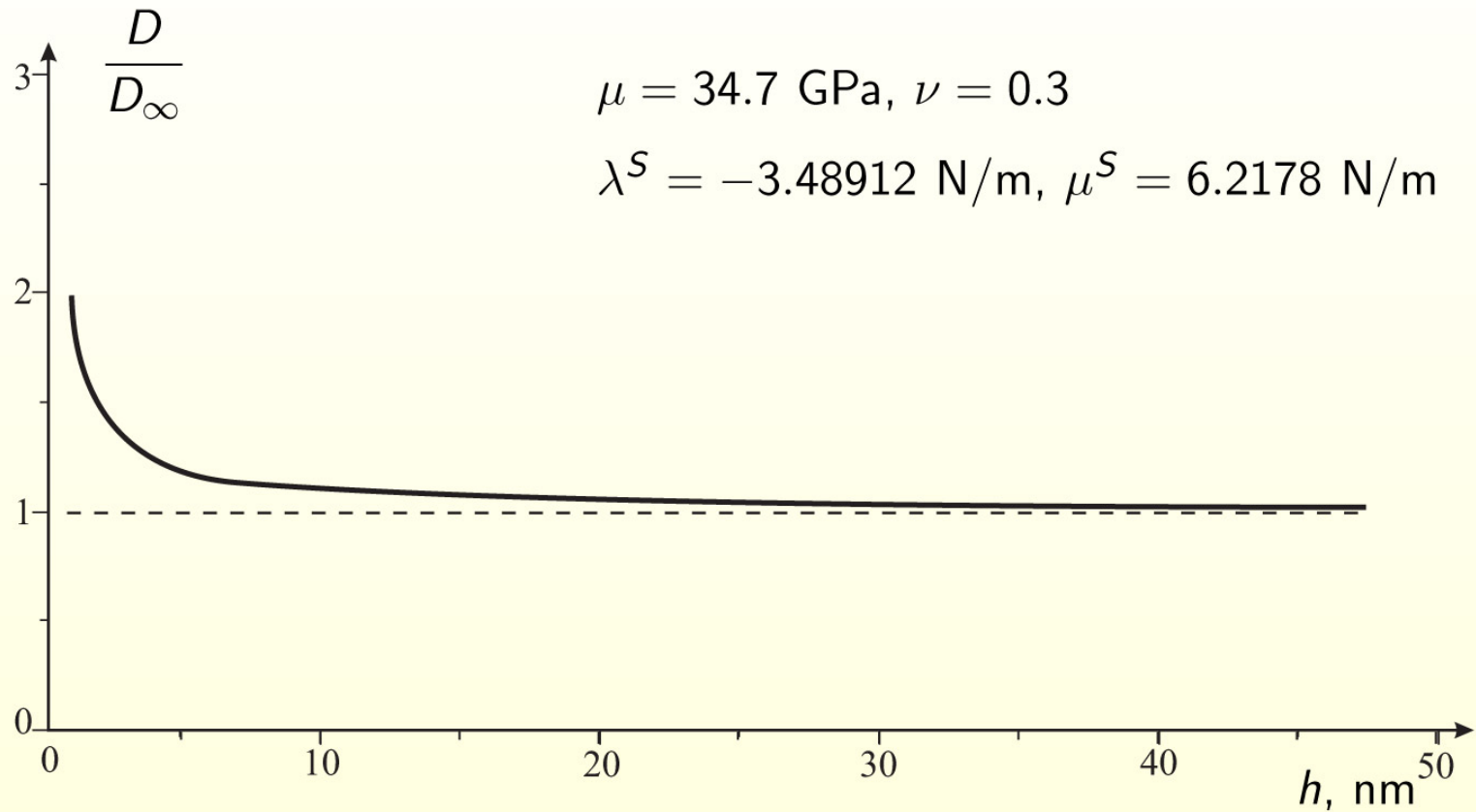
$$D \equiv D_1 + D_2 = D_\infty + D_{\text{surface}},$$
$$D_\infty = \frac{Eh^3}{12(1-\nu^2)}, \quad D_{\text{surface}} = h^2(\mu^S + \lambda^S/2)$$

From the positive definiteness of the surface energy density it follows

$$\mu_S > 0, \quad \mu_S + \lambda_S > 0 \quad \implies \quad D_{\text{surface}} > 0$$



Bending Stiffness of a Plate made of Al⁶

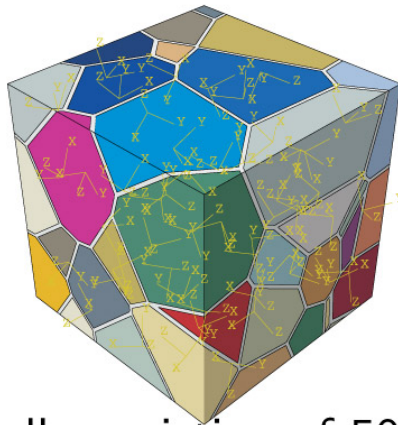


⁶Duan et al. (2008)

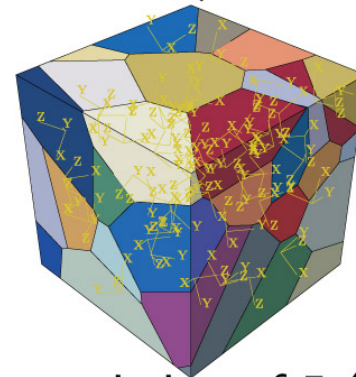
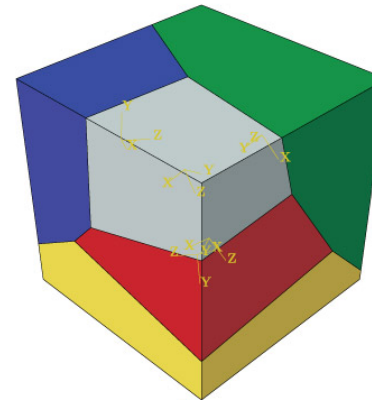
Geometrical Model of a Polycrystal

Main User Defined Parameters

- Number of grains
- Dimensions of the unit cell
- Average grain size
- Grain border thickness



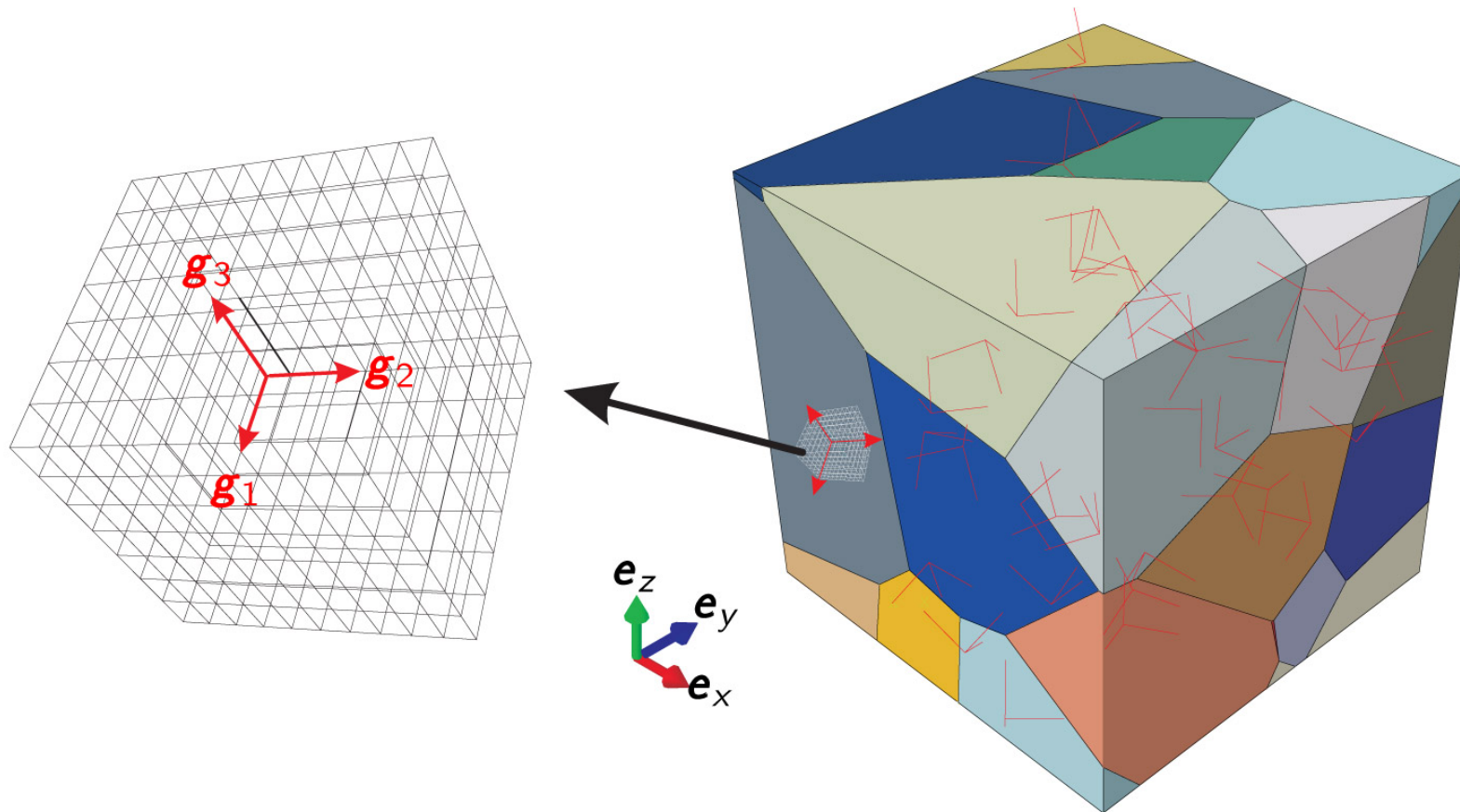
Unit cell consisting of 50 grains
with non-zero grain border thickness



Unit cells consisting of 5 (50) grains
with zero grain border thickness

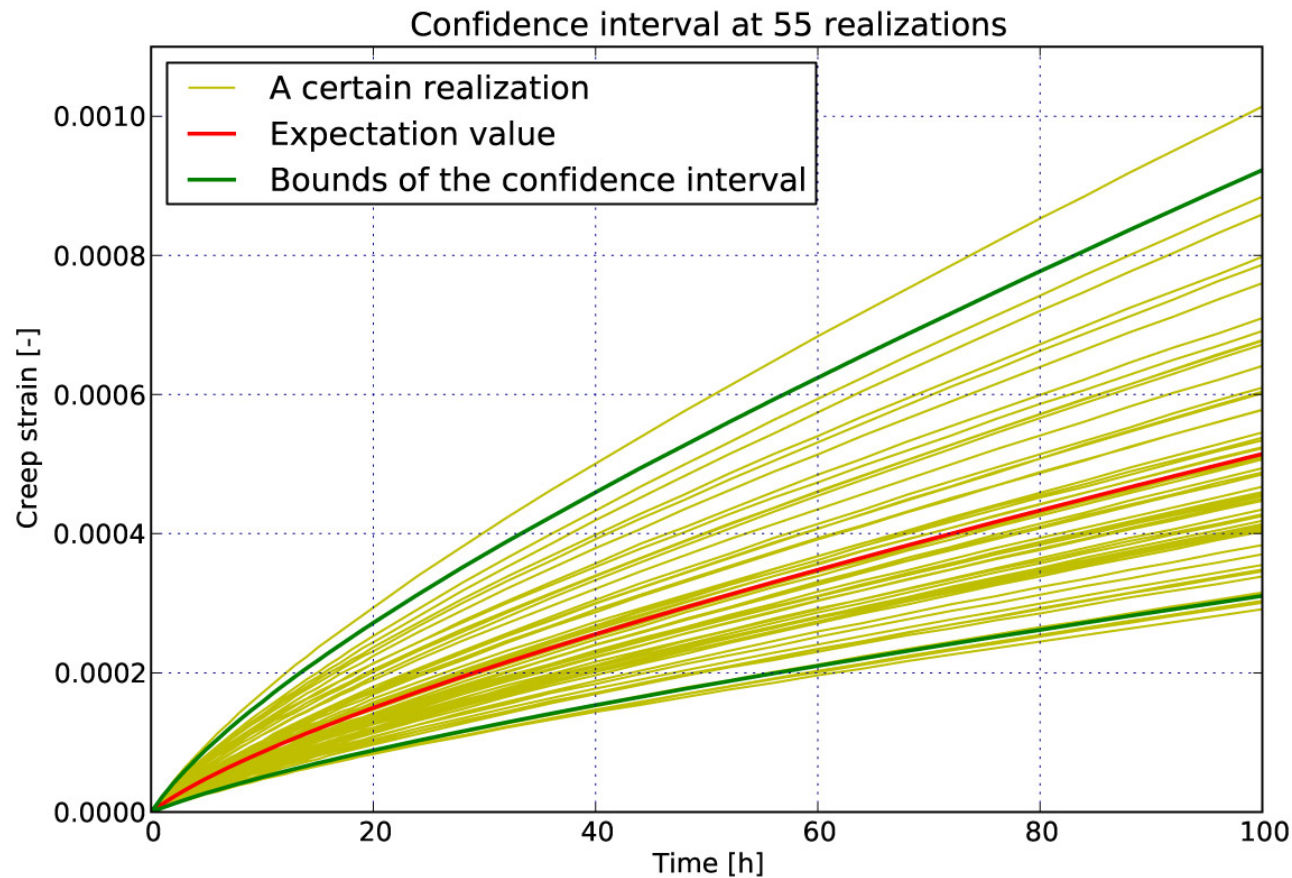
Modeling on Different Scales

Model of Polycrystal



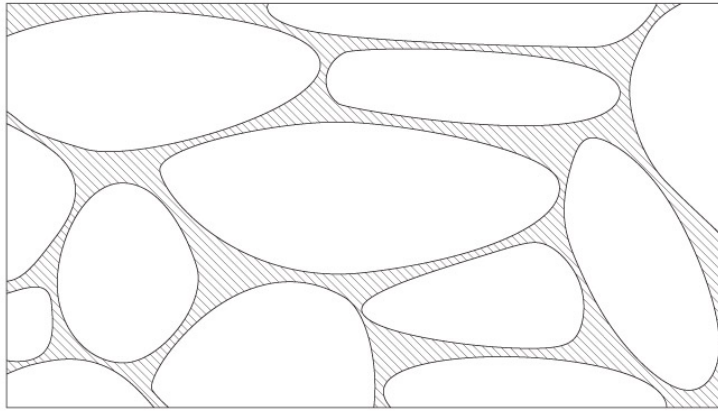
Creep Curves of 55 Realizations

Bounds of the Confidence Interval and Expectation Value

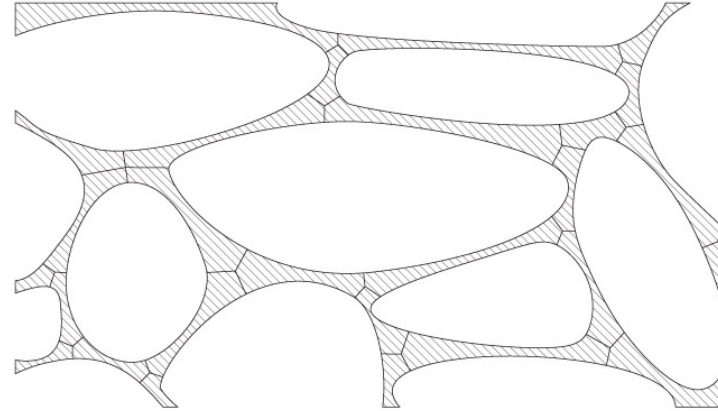


Porous Material Representation

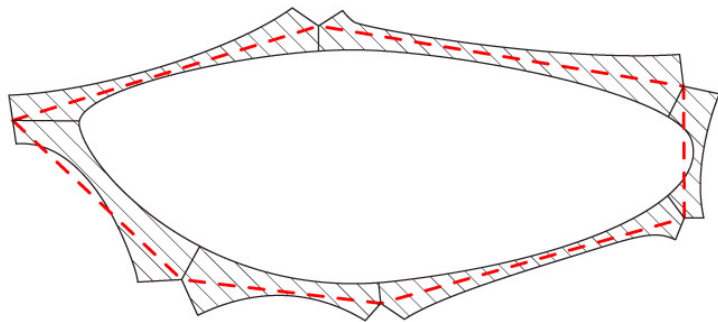
Closed Cell Foam Scheme



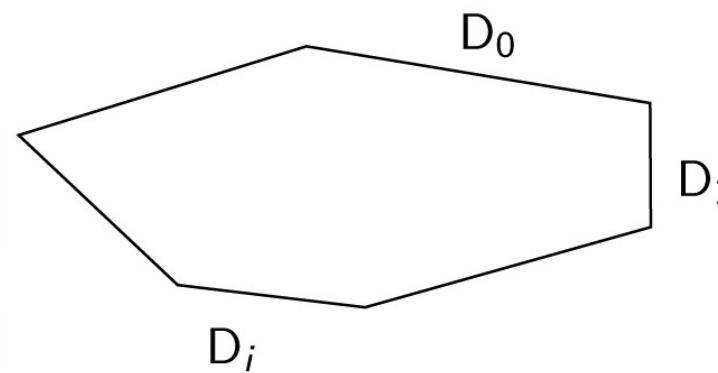
Interconnected Shells



Foam Cell



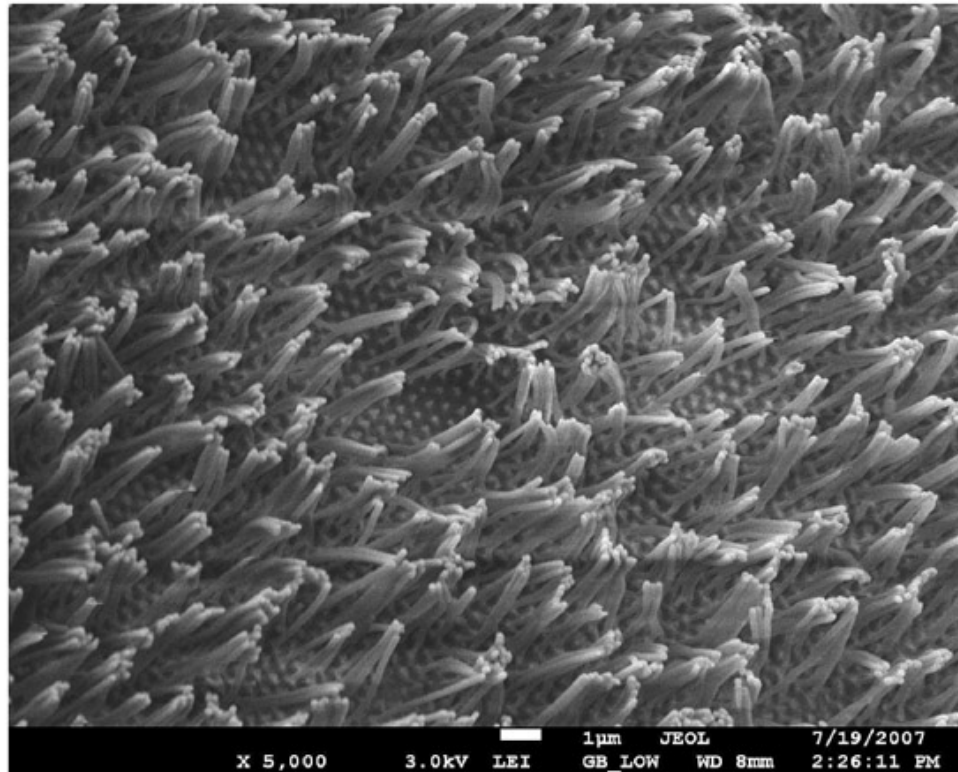
2D Voronoi Tessellation



Disordered Nanostructure

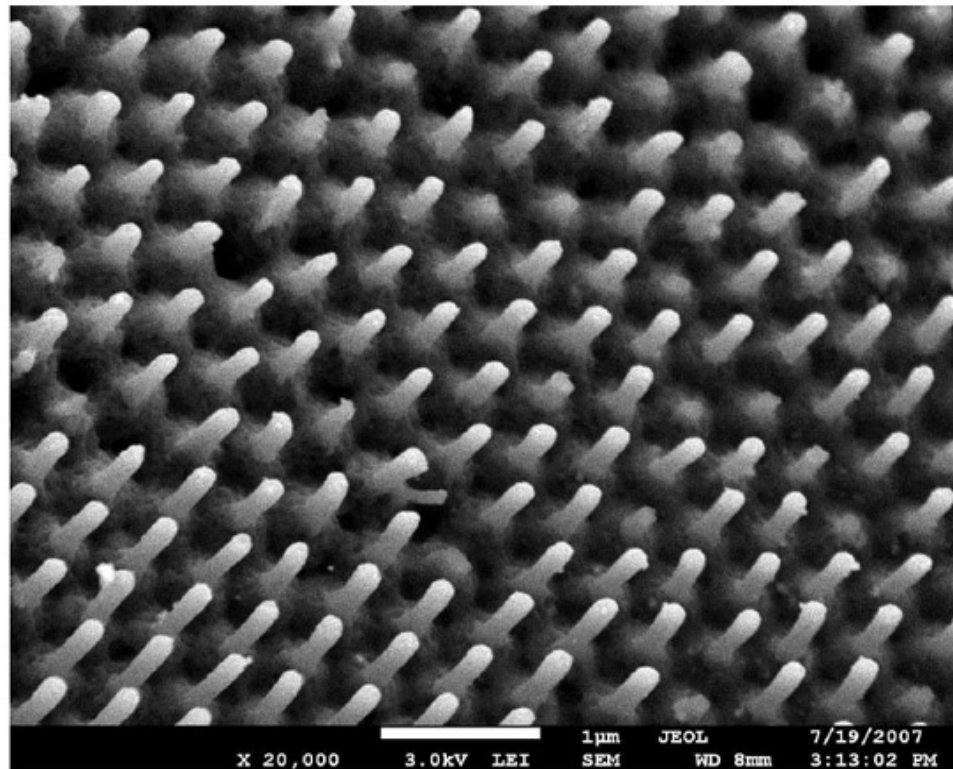
Crosslinked Polyetheracrylate: diameter: 120 nm, length: 4 μm

Disordered nanostructure as result of what:
the loss of stability (weight?) or adhesive forces



Ordered Nanostructure

Crosslinked Polyetheracrylate: diameter: 120 nm, length: 1 μm



Thank you for your attention!

Further questions:
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